# SIMULATION OF POROUS LOW-*k* DIELECTRIC SEALING BY COMBINED He AND NH<sub>3</sub> PLASMA TREATMENT<sup>\*</sup>

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## AGENDA

- Low-*k* Dielectrics
- Modeling Platforms
- Modeling of Porous Low-*k* Sealing
  - Goals and Premises for Sealing Mechanism
- Sealing Mechanism
  - Surface Site Activation by He plasma pre-treatment
  - Sealing by Ar/NH<sub>3</sub> Treatment
- Sealing Efficiency Dependence
  - Porosity and Interconnectivity
  - Treatment time and Pore Radius
- Concluding Remarks

# **POROUS LOW-***k* **DIELECTRIC**

- Metal interconnect lines in ICs run through dielectric insulators.
- The capacitance of the insulator contributes to *RC* delays.
- Porous oxides, such as C doped SiO<sub>2</sub> (with CH<sub>n</sub> lining pores) have a low dielectric constant which reduces the RC delay.
- Porosity is ≤ 0.5. Interconnected pores open to surface offer pathways to degrade *k*-value by reactions.



Ref: http://www.necel.com/process/en/images/porous\_low-k\_e.gif



#### **GOALS AND PREMISES OF SEALING MECHANISM**

- To prevent the degradation of lowk materials pores open to the surface has to be sealed.
- He followed by NH<sub>3</sub> plasma treatment has been shown to seal the pores.
  - He<sup>+</sup> and photons break Si-O bonds while knocking off H atom from CH<sub>n</sub>.

Plasma	Treatment Time (s)	Function
He	20	Surface Activation
NH <sub>3</sub>	20 ( Post-He)	Sealing

- Subsequent NH<sub>3</sub> exposure seals the pores by adsorption reactions forming C-N and Si-N bonds.
- Experimental results from the literature were used to build the sealing mechanism.

Ref: A. M. Urbanowicz, M. R. Baklanov, J. Heijlen, Y. Travaly, and A. Cockburn, Electrochem. Solid-State Lett. 10, G76 (2007).

#### **MODELING : LOW-***k* **PORE SEALING**



- Hybrid Plasma
  Equipment Model (HPEM)
- Plasma Chemistry Monte Carlo Module (PCMCM)
- Monte Carlo Feature Profile Model (MCFPM)

# HYBRID PLASMA EQUIPMENT MODEL (HPEM)



Carlo Radiation • SCM (Surface Transport Module) Chemistry Mo

Chemistry Module) University of Michigan Institute for Plasma Science & Engr.

#### MONTE CARLO FEATURE PROFILE MODEL (MCFPM)



- The MCFPM resolves the surface topology on a 2D Cartesian mesh to predict etch profiles.
- Each cell in the mesh has a material identity. (Cells are 4 x 4 A).
- Gas phase species are represented by Monte Carlo pseudoparticles.
- Pseuodoparticles are launched towards the wafer with energies and angles sampled from the distributions obtained from the PCMCM.
- Cells identities changed, removed, added for reactions, etching deposition.

#### **INITIAL LOW-***k* **PROFILE FOR SIMULATION**



- 80 nm wide and 30 nm thick porous SiO<sub>2</sub>
- CH<sub>3</sub> groups line the pores
- Average pore radius: 0.8-1.4 nm
- Pores open to surface need to be sealed
- Will be exposed to successive He and NH<sub>3</sub> plasmas.

## SURFACE ACTIVATION IN He PLASMA

- He<sup>+</sup> and photons break Si-O bonds and removes H from CH<sub>3</sub> groups.
- $\begin{array}{lll} \bullet \mbox{ Bond Breaking } & \mbox{He}^{*}(g) + SiO_{2}(s) \rightarrow SiO(s) + O(s) + \mbox{He}(g) \\ & \mbox{He}^{*}(g) + SiO(s) \rightarrow Si(s) + O(s) + \mbox{He}(g) \\ & \mbox{hv} + SiO_{2}(s) & \rightarrow SiO(s) + O(s) \\ & \mbox{hv} + SiO(s) & \rightarrow Si(s) + O(s) \\ & \mbox{hv} + SiO(s) & \rightarrow Si(s) + O(s) \\ & \mbox{He}^{*}(g) + CH_{n}(s) & \rightarrow CH_{n-1}(s) + \mbox{H}(g) + \mbox{He}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) + \mbox{He}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \rightarrow CH_{n-2}(s) + \mbox{H}(g) \\ & \mbox{hv} + CH_{n-1}(s) & \mbox{H}(g) \\ & \mbox{hv} + CH_{n-2}(s) + \mbox{hv} + \m$
- Reactive sites assist sealing in the subsequent Ar/NH<sub>3</sub> treatment.

### **SEALING MECHANISM IN Ar/NH<sub>3</sub> PLASMA**

- N/NH<sub>x</sub> species are adsorbed by activated sites forming Si-N and C-N bonds to seal pores.
- Further Bond Breaking  $M^+(g) + SiO_2(s) \rightarrow SiO(s) + O(s) + M(g)$ 
  - $M^+(g) + SiO(s) \rightarrow Si(s) + O(s) + M(g)$
- - $NH_x(g) + CH_n(s) \rightarrow CH_nNH_x(s)$

 $NH_x(g) + C(s) \rightarrow CNH_x(s)$ 

 SiNH<sub>x</sub>-NH<sub>y</sub>/CNH<sub>x</sub>-NH<sub>y</sub> compounds help seal the pores where end nitrogens are bonded to either Si or C atom by Si-C/Si-N bond

 $NH_{v}(g) + SiNH_{x}(s) \rightarrow SiNH_{x}-NH_{v}(s)$ 

 $NH_y(g) + CNH_x(s) \rightarrow CNH_x-NH_y(s)$ 

- He<sup>+</sup> and photons in He plasma break Si-O bonds and activate CH<sub>n</sub> groups.
- He Plasma Species:

He He\* He\* hv e

- $Ar/NH_3 = 25/75$  treatment seals the surface pores.
- Ar/NH<sub>3</sub> Plasma Species:

Ar	Ar*	Ar <sup>+</sup>	е	
NH <sub>3</sub>	$NH_2$	NH	Н	Ν
NH <sub>3</sub> <sup>+</sup>	$NH_2^+$	$NH_4^+$	NH <sup>+</sup>	

- lon density: 3.8 x 10<sup>10</sup> cm<sup>-3.</sup>
- Porous low-k was exposed for 30s to the plasma.
- 20V substrate bias assisted ablating H and Si-O bond breaking.
- Conditions: He, 10 mTorr, 300 W ICP, 20V Bias



# Ar/NH<sub>3</sub> PLASMAS

- Total ion density: 1.0x 10<sup>11</sup> cm<sup>-3</sup>
- Ion densities (cm<sup>-3</sup>):  $NH_3^+ 2.6 \times 10^{10}$   $NH_4^+ 2.9 \times 10^{10}$   $NH_2^+ 1.0 \times 10^{10}$   $NH^+ 1.4 \times 10^{09}$  $H^+ 1.6 \times 10^{10}$
- Neutral densities (cm<sup>-3</sup>):

NH <sub>3</sub>	5.30 x 10 <sup>13</sup>
NH <sub>2</sub>	<b>2.40 x 10</b> <sup>13</sup>
NH	1.6 x 10 <sup>12</sup>
Ν	<b>2.4</b> x 10 <sup>12</sup>
Ar	6.0 x 10 <sup>12</sup>



Conditions: Ar/NH<sub>3</sub> = 25/75, 10 mTorr, 300 W ICP

# PORE-SEALING BY SUCCESSIVE He AND NH<sub>3</sub>/Ar TREATMENT



- Surface pore sites are activated by 30s He plasma treatment.
- Successive 20s NH<sub>3</sub> treatment seals the pores forming Si-N and Si-C bonds.

Animation Slide-GIF

## SEALING: POROSITY AND INTERCONNECTIVITY

- Sealing efficiency is independent of porosity and interconnectivity, optimizing at 75-80%
- With higher porosity, the number of open pores to the surface increases.
- If pore radius remains the same, sealing efficiency is constant.
- With higher porosity but a fixed pore radius, number of surface pores increases.
- The fixed probabilities of C-N, Si-N and N-N bond formation result in a constant sealing efficiency.



### **SEALING: TREATMENT TIME DEPENDENCE**

- Without He plasma treatment, Ar/NH<sub>3</sub> plasmas seal only 45% of pores.
- NH<sub>x</sub> ions are unable to activate all the surface sites to complete the sealing.
- Sealing efficiency increases with He treatment time for 30s, then saturates.
- 30s treatment breaks all surface Si-O bonds and activates all surface CH<sub>3</sub> groups.
- Sealing efficiency of pores increases for 20s of Ar/NH<sub>3</sub> treatment, then saturates – all dangling bonds on the surface are passivated.



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#### **SEALING: He TREATMENT TIME DEPENDENCE**

- He plasma is responsible for Si-O bond breaking and removing H from CH<sub>3</sub> groups to create reactive sites.
- Increasing He plasma treatment time increases sealing efficiency until all of the surface sites are activated.



# **SEALING: Ar/NH<sub>3</sub> TREATMENT TIME DEPENDENCE**



- NH<sub>x</sub> species are adsorbed by reactive sites produced by He plasma to form Si-C and Si-N bonds.
- 80% of surface pores are sealed within 20s...all surface activated sites are passivated by C-N/Si-N bonds.

**Animation Slide-GIF** 



# **SEALING EFFICIENCY: PORE RADIUS**

- Sealing efficiency decreases with increasing pore size.
- Sealing efficiency drops below 70% as for pore radius > 1.0 nm.
- C-N and Si-N are "first bonds."
- Sealing requires N-N bonding, which has limited extent.
- Too large a gap prevents sealing.



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#### **CONCLUDING REMARKS**

- Simulation of porous low-k material sealing was investigated employing successive He and NH<sub>3</sub> plasma treatment.
- Si-N and C-N bonds formed by adsorption on active sites followed by one N-N bond linking C or Si atoms from opposite pore walls.
- Pore sealing efficiency is independent of porosity and interconnectivity, while dependent on both He and NH<sub>3</sub> plasma treatment time.
- The sealing efficiency degrades when the pore radius is greater than 1 nm.
- Sealing efficiency will improve if the pore radius standard deviation can be maintained low.