

# **PLASMA PROCESSING AND POLYMERS: FRITO BAGS TO MICROELECTRONICS FABRICATION**

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**November 2002**

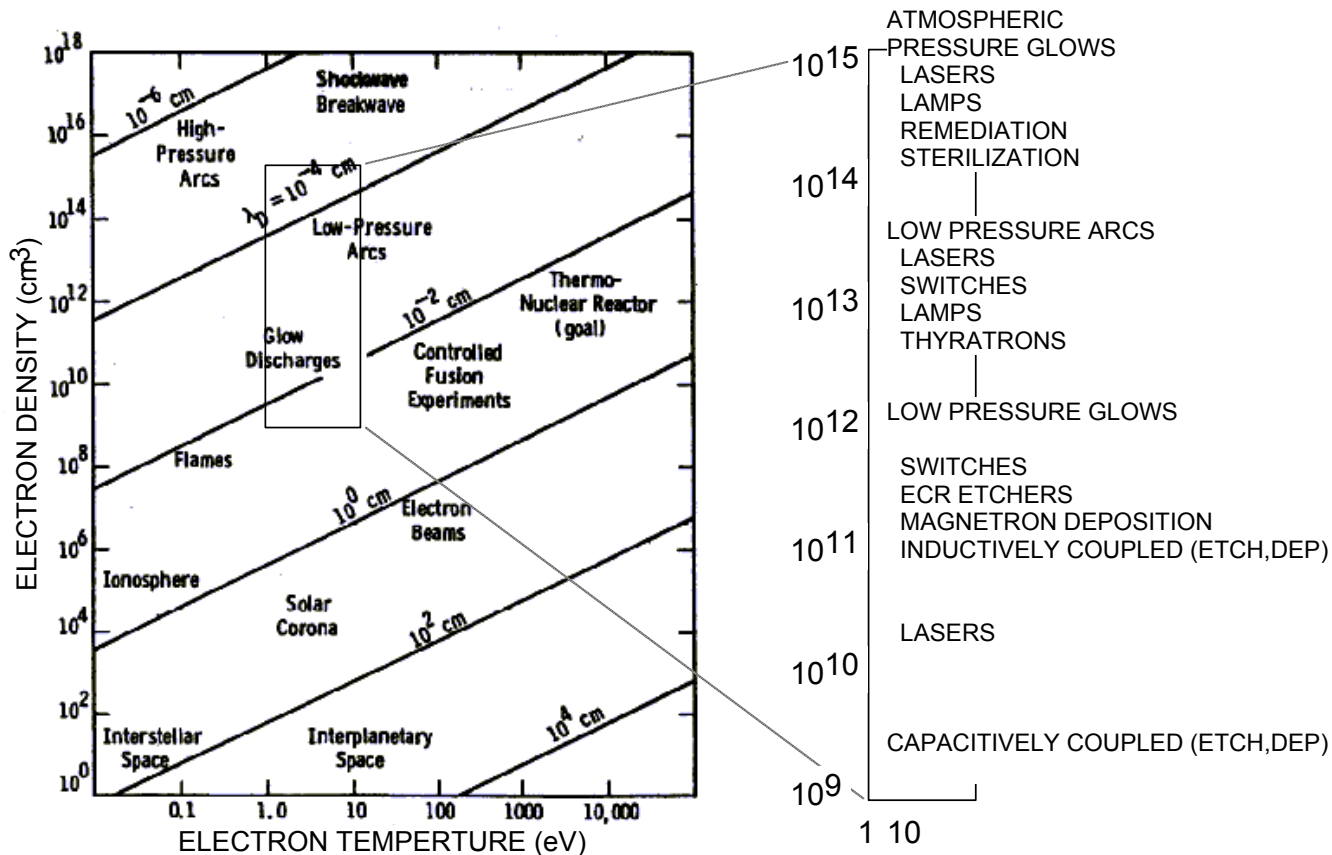
# AGENDA

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- **Plasmas: Tools for eV physics and chemistry**
- **Plasmas and Polymers: Extremes in Physics and Applications**
- **Plasmas for functionalization of polymers**
- **Polymers for selectivity in plasma etching**
- **Concluding Remarks**

# PARTIALLY IONIZED PLASMAS

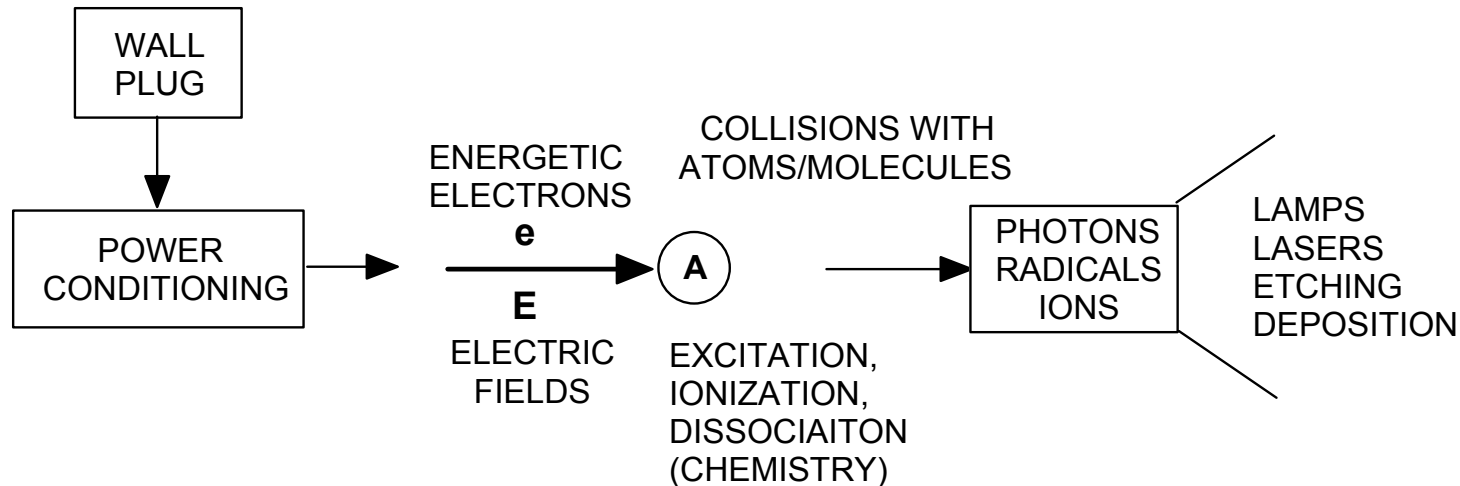
- Partially ionized plasmas are gases containing neutral atoms and molecules, electrons, positive ions and negative ions.
- An air plasma:  $N_2$ ,  $O_2$ ,  $N_2^+$ ,  $O_2^+$ ,  $O^-$ ,  $e$  where  $[e] \ll$  Neutrals



# COLLISIONAL LOW TEMPERATURE PLASMAS

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- These systems are the plasmas of every day technology.

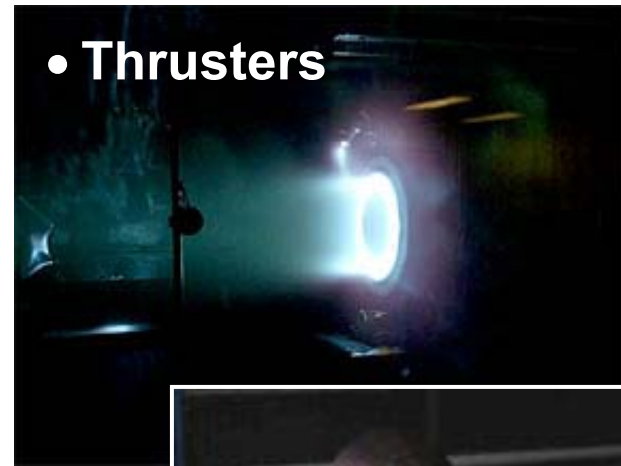


- Electrons transfer power from the "wall plug" to internal modes of atoms / molecules to "make a product", very much like combustion.
- The electrons are "hot" (several eV or 10-30,000 K) while the gas and ions are cool, creating "non-equilibrium" plasmas

# COLLISIONAL LOW TEMPERATURE PLASMAS



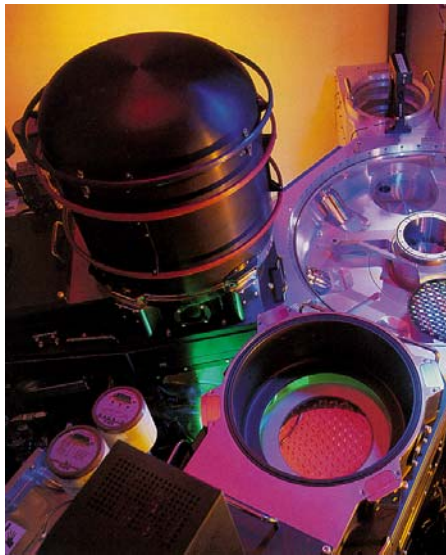
• Lighting



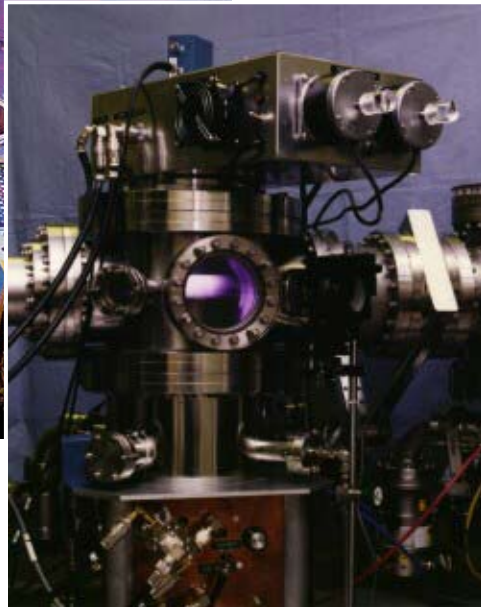
• Thrusters



• Spray Coatings



• Materials Processing

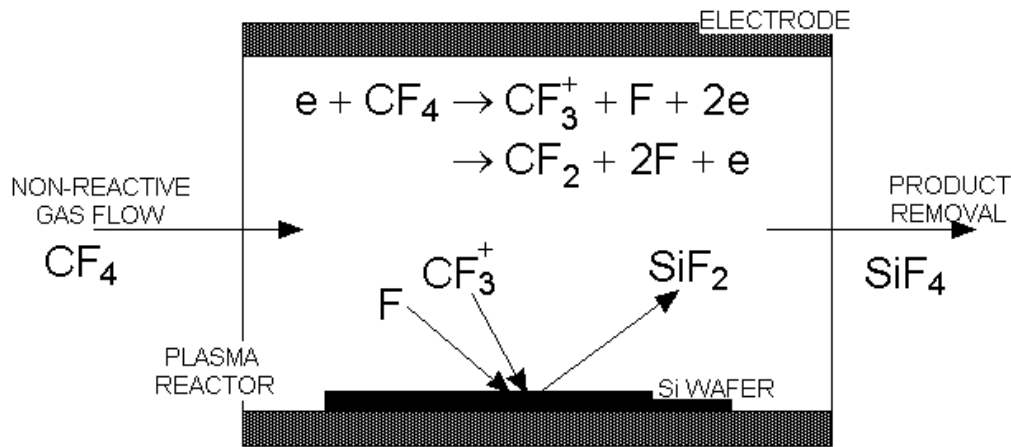


• Displays

# PLASMAS FOR MODIFICATION OF SURFACES

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- Plasmas are ideal for producing reactive species (radicals, ions) for modifying surface properties.
- Two of the most technologically (and commercially) important uses of plasmas involve polymers:



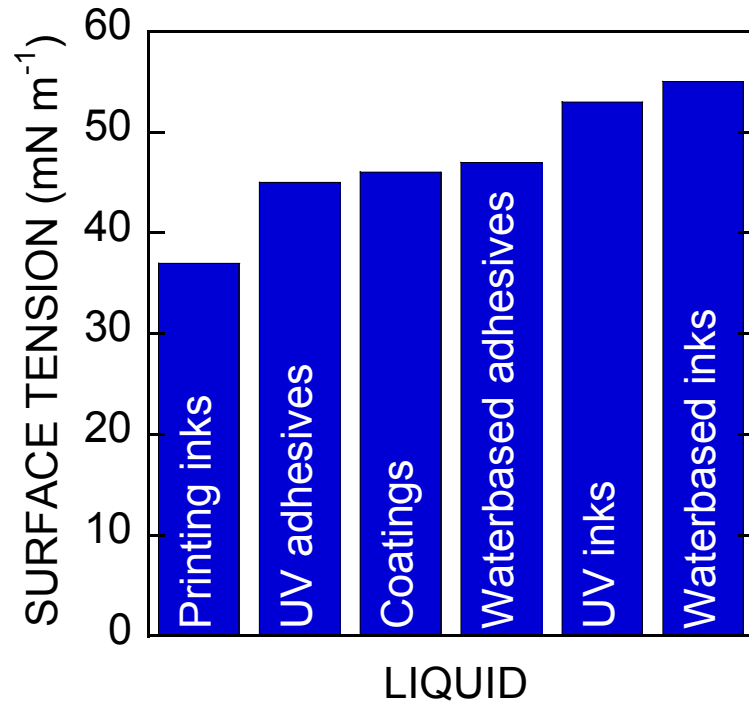
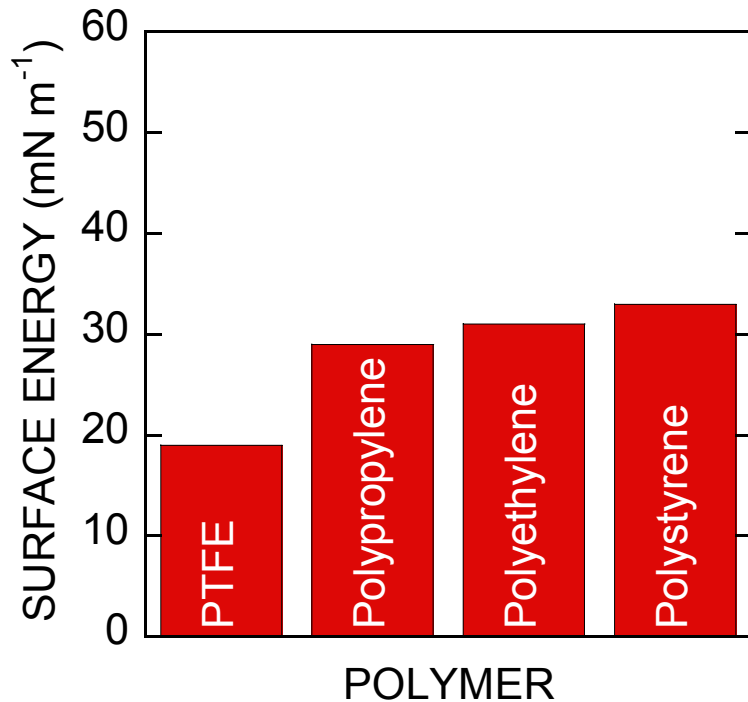
- Functionalization of surfaces (high pressure)
- Etching for microelectronics fabrication (low pressure)

- Both applications utilize unique properties of low temperature plasmas to selectively produce structures.

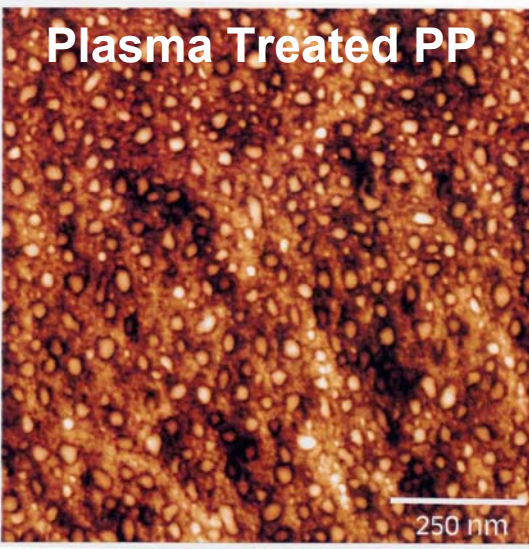
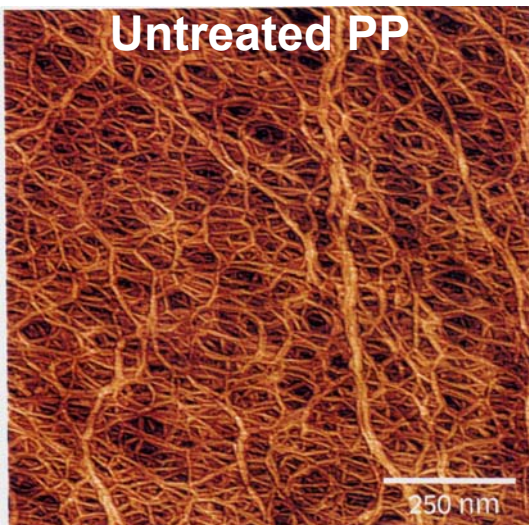
# SURFACE ENERGY AND FUNCTIONALITY OF POLYMERS

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- Most polymers, having low surface energy, are hydrophobic.
- For good adhesion and wettability, the surface energy of the polymer should exceed of the overlayer by  $\approx 2-10 \text{ mN m}^{-1}$ .



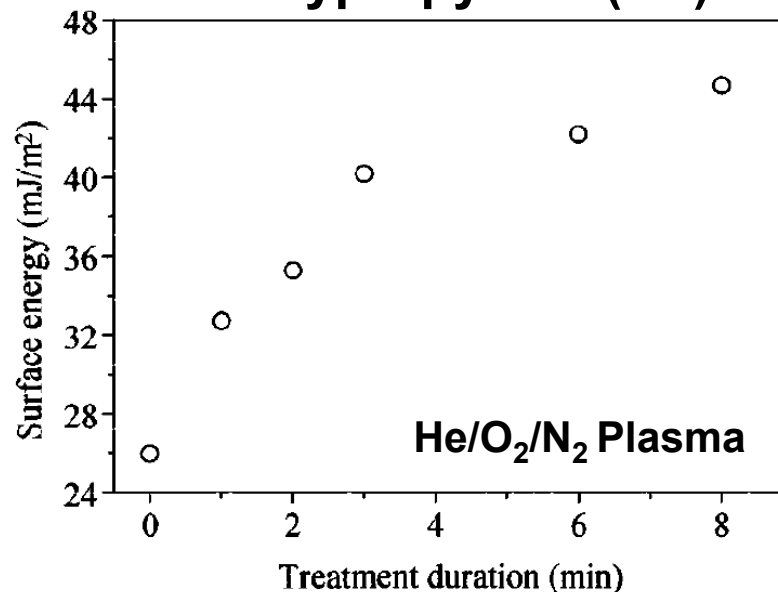
# PLASMA SURFACE MODIFICATION OF POLYMERS



• M. Strobel, 3M

- To improve wetting and adhesion of polymers atmospheric plasmas are used to generate gas-phase radicals to functionalize their surfaces.

## • Polypropylene (PP)

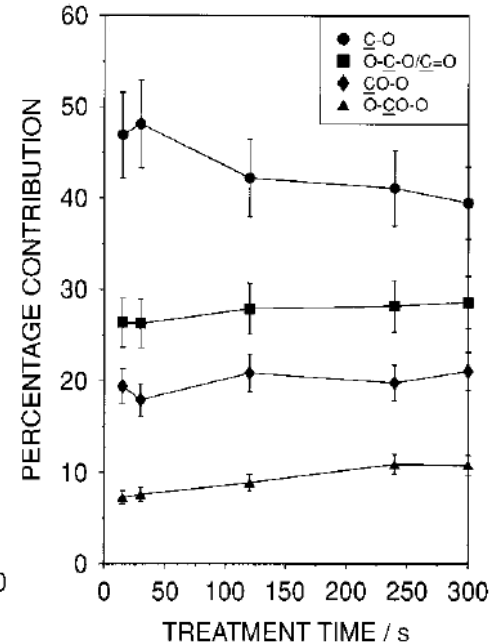
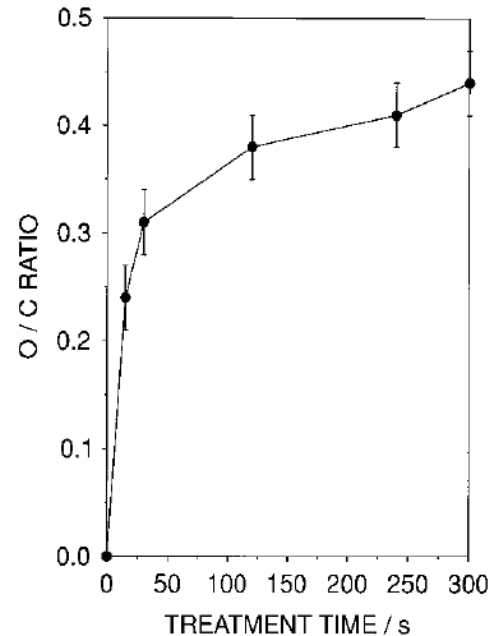
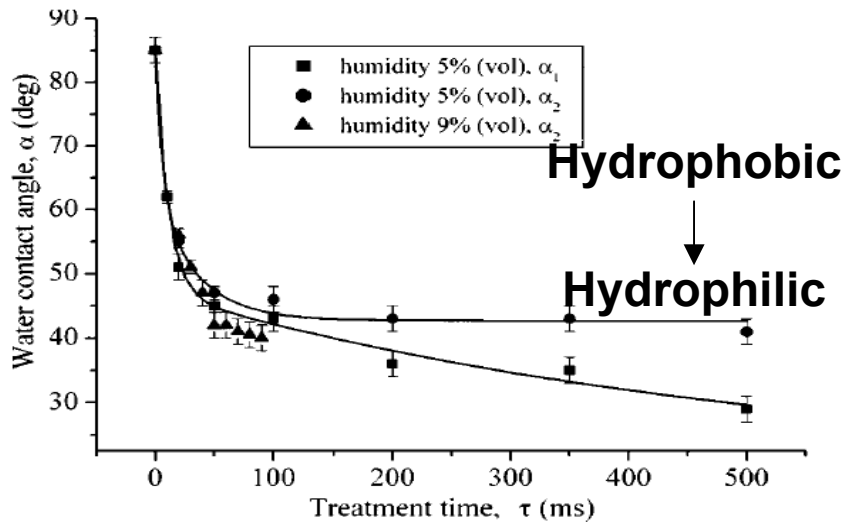


- Massines *et al.* J. Phys. D 31, 3411 (1998).



# PLASMA PRODUCED WETTABILITY

- Increases in wettability with plasma treatment result from formation of surface hydrophilic groups such as C-O-O (peroxy), C=O (carbonyl).



- Polyethylene, Humid-air
- Akishev, Plasmas Polym. 7, 261 (2002).

- Polypropylene, Air corona
- Boyd, Macromol., 30, 5429 (1997).

# POLYMER TREATMENT APPARATUS

- TYPICAL PROCESS CONDITIONS:

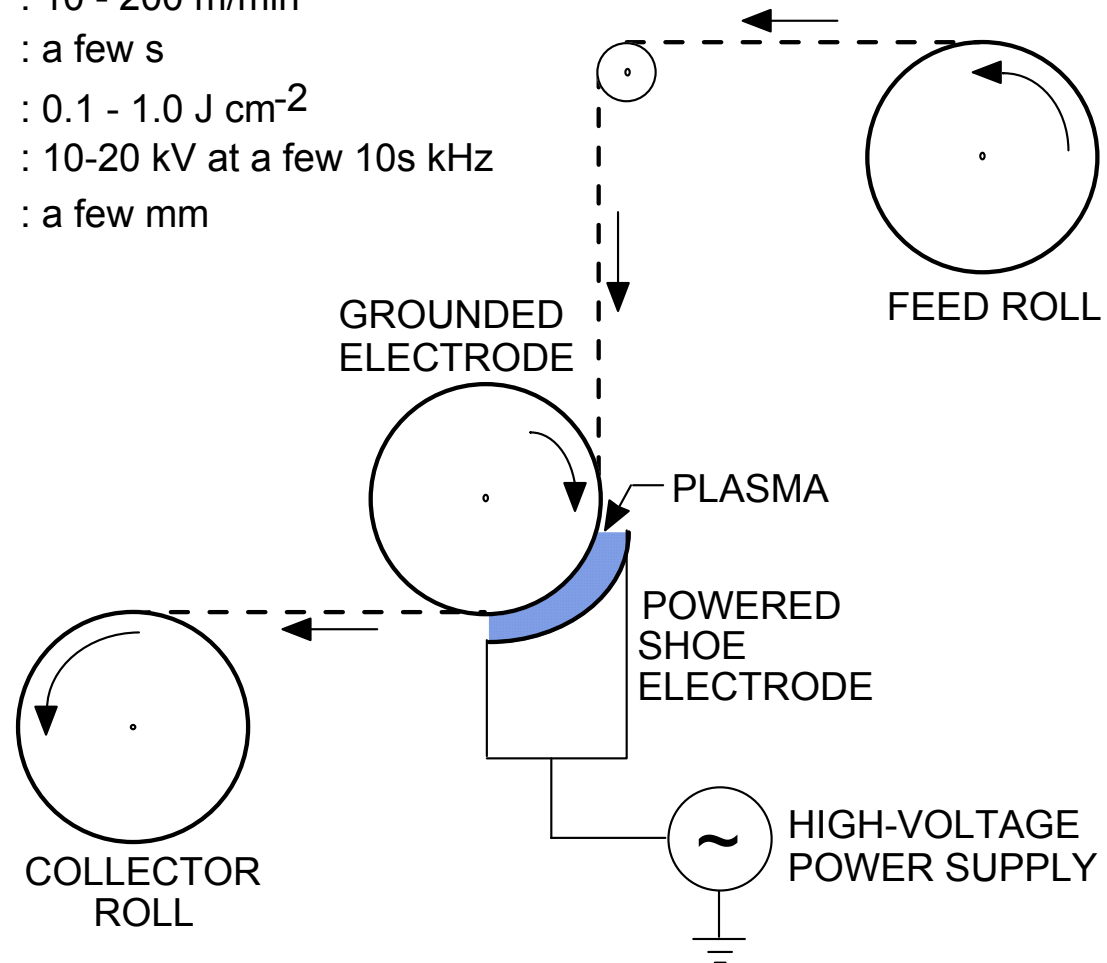
Web speed : 10 - 200 m/min

Residence time : a few s

Energy deposition : 0.1 - 1.0 J cm<sup>-2</sup>

Applied voltage : 10-20 kV at a few 10s kHz

Gas gap : a few mm



# COMMERCIAL CORONA PLASMA EQUIPMENT

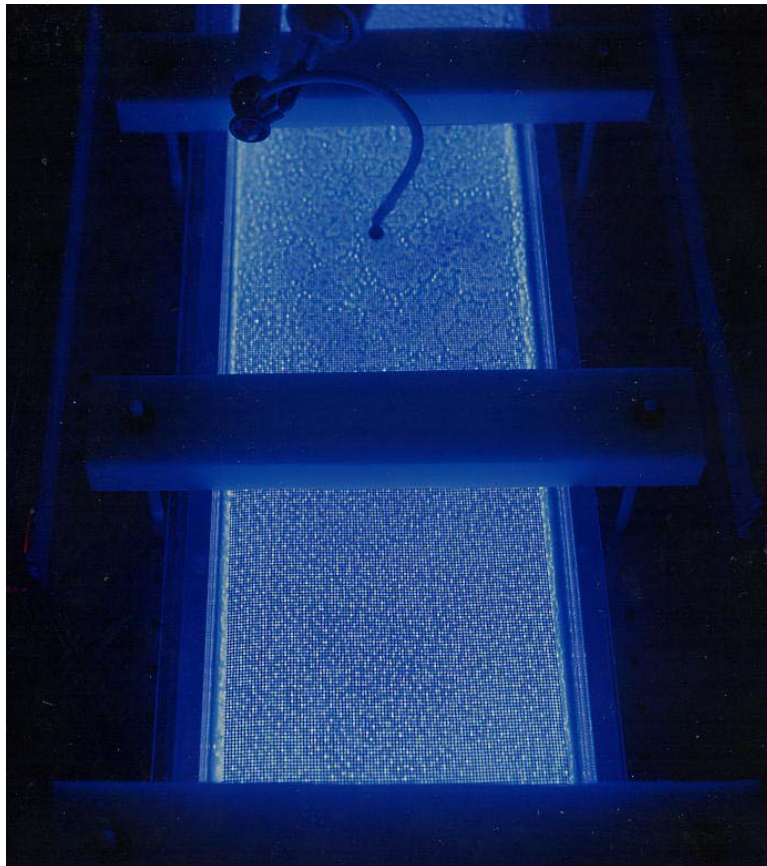
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Tantec Inc.

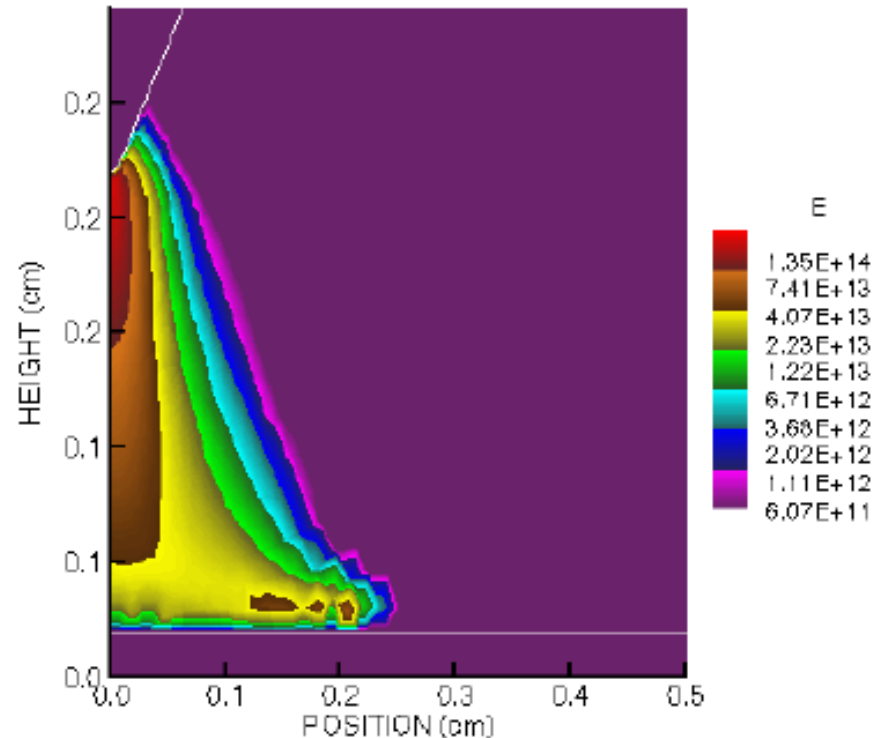
# CORONA/DIELECTRIC BARRIER PLASMAS

- Corona and dielectric barrier discharge plasmas operate in a filamentary mode.



- **Laboratory Dielectric Barrier Discharge**

- **Electron Density (2-d Model)**

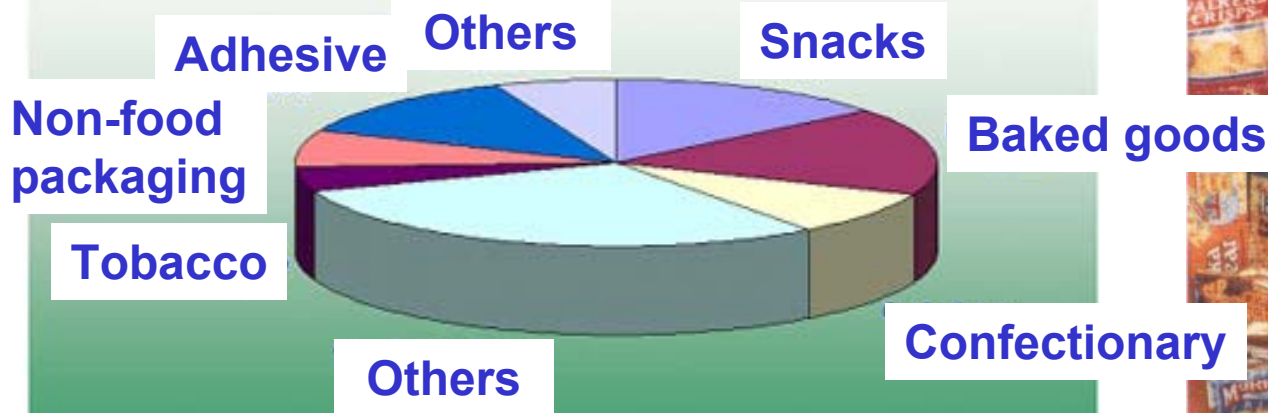


- **1 atm, Dry Air, -15 kV, 30 ns**

# POLYPROPYLENE

- PP is a hard but flexible plastic. 5 million metric tons of PP film are used yearly, much of it functionalized with plasmas

Worldwide market of BOPP by end uses



Source : PCI Films Consulting Ltd.

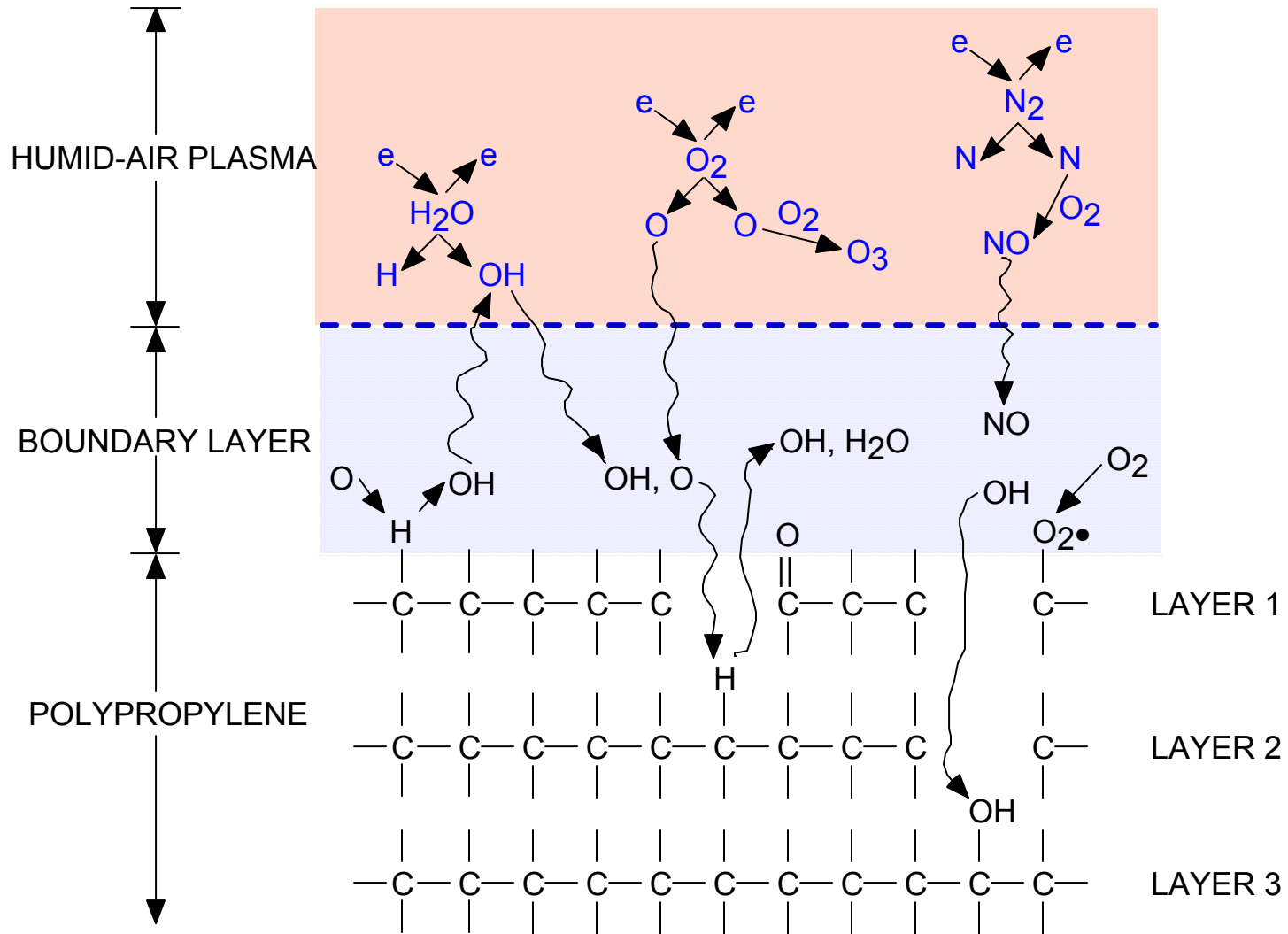


# FUNCTIONALIZATION OF THE PP SURFACE

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- Untreated PP is hydrophobic.
- Increases in surface energy by plasma treatment are attributed to the functionalization of the surface with hydrophilic groups.
  - Carbonyl ( $\text{-C=O}$ )
  - Alcohols ( $\text{C-OH}$ )
  - Peroxy ( $\text{-C-O-O}$ )
  - Acids ( $\text{((OH)C=O)}$ )
- The degree of functionalization depends on process parameters such as gas mix, energy deposition and relative humidity (RH).
- At sufficiently high energy deposition, erosion of the polymer occurs.

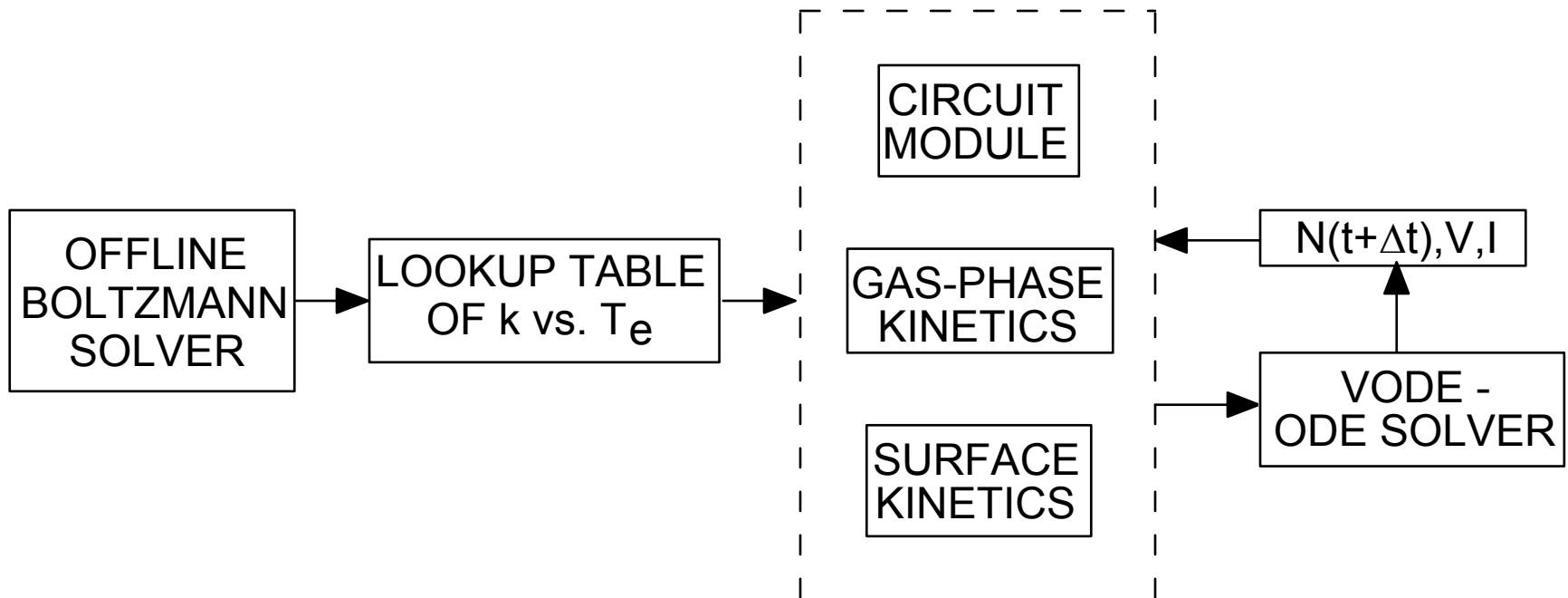
# REACTION PATHWAY



# DESCRIPTION OF THE MODEL: GLOBAL\_KIN

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- Modules in GLOBAL\_KIN:
  - Circuit model
  - Homogeneous plasma chemistry
  - Species transport to PP surface
  - Heterogeneous surface chemistry

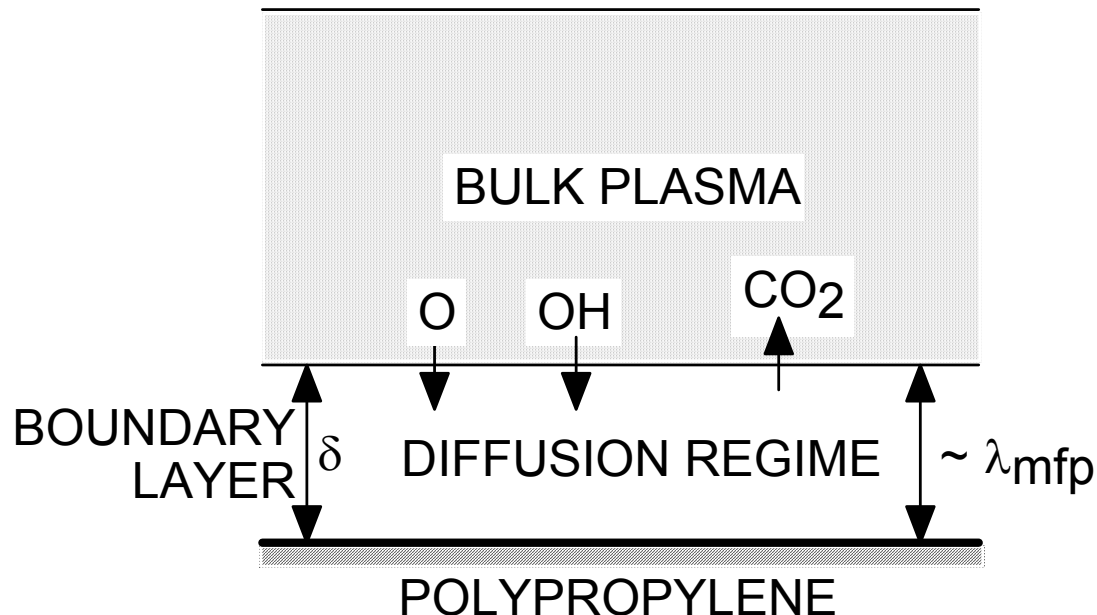




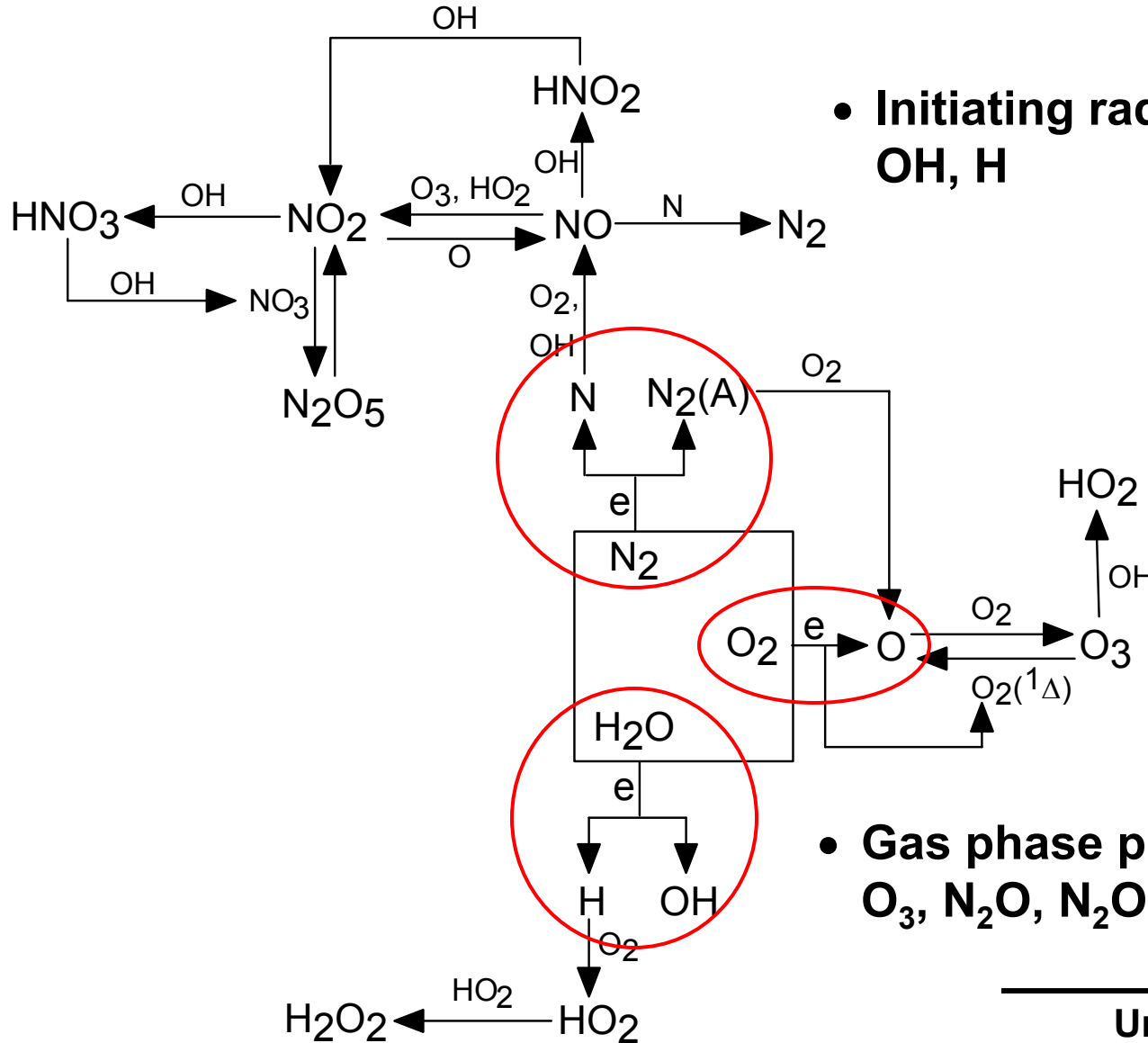
# SPECIES TRANSPORT TO THE POLYMER SURFACE

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- Species in the bulk plasma diffuse to the PP surface through a boundary layer ( $d \sim$  a few  $\lambda_{\text{mfp}} \approx \mu\text{m}$ ).
- Radicals react on the PP based on a variable density, multiple layer surface site balance model.



# REACTION MECHANISM FOR HUMID-AIR PLASMA

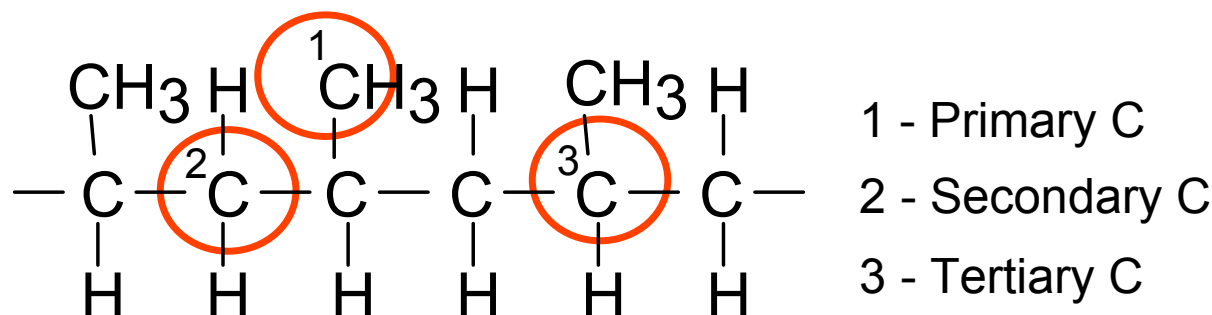


- Initiating radicals are O, N, OH, H

- Gas phase products include  $O_3$ ,  $N_2O$ ,  $N_2O_5$ ,  $HNO_2$ ,  $HNO_3$ .

# POLYPROPYLENE (PP) POLYMER STRUCTURE

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- Three types of carbon atoms in a PP chain:
  - Primary – bonded to 1 C atom
  - Secondary – bonded to 2 C atoms
  - Tertiary – bonded to 3 C atoms
- The reactivity of an H-atom depends on the type of C bonding.  
Reactivity scales as:

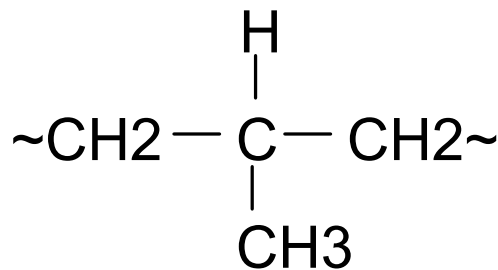


# PP SURFACE REACTION MECHANISM: INITIATION

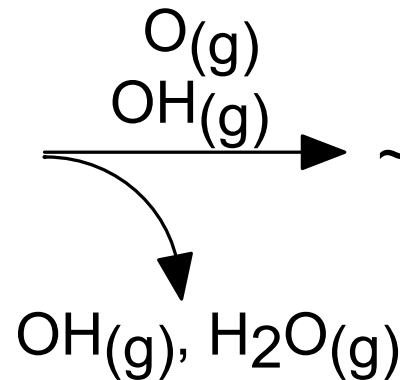
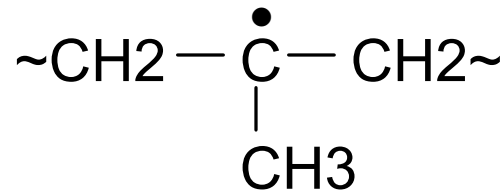
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- The surface reaction mechanism has *initiation*, *propagation* and *termination* reactions.
- **INITIATION**: O and OH abstract H from PP to produce alkyl radicals; and gas phase OH and H<sub>2</sub>O.

(POLYPROPYLENE)



(ALKYL RADICAL)

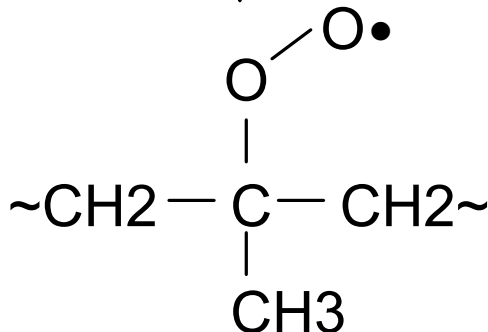
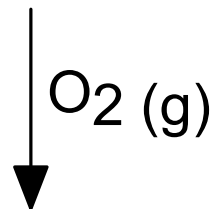
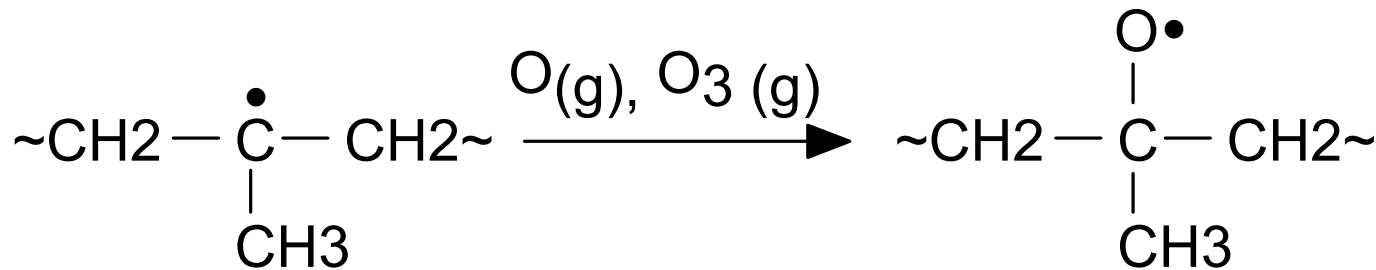


# PP SURFACE REACTION MECHANISM: PROPAGATION

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(ALKYL RADICAL)

(ALKOXY RADICAL)

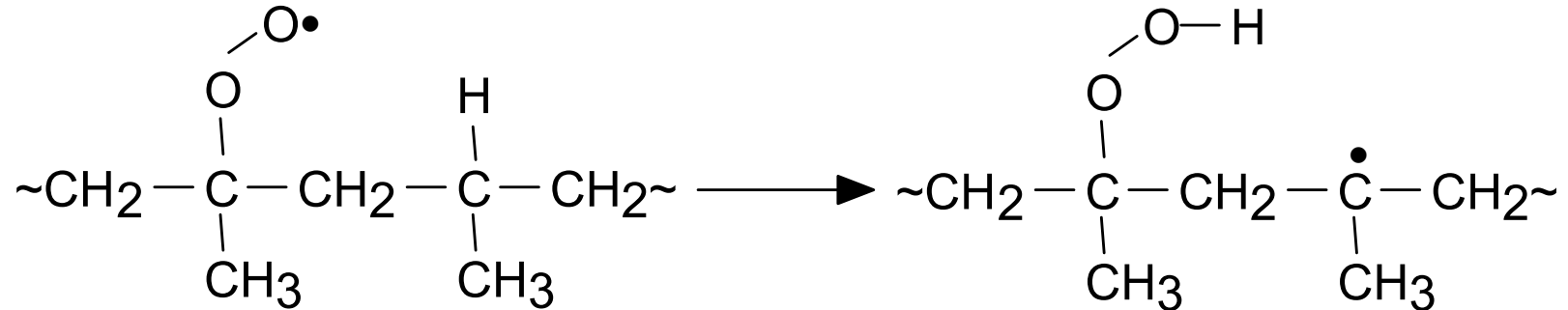


(PEROXY RADICAL)

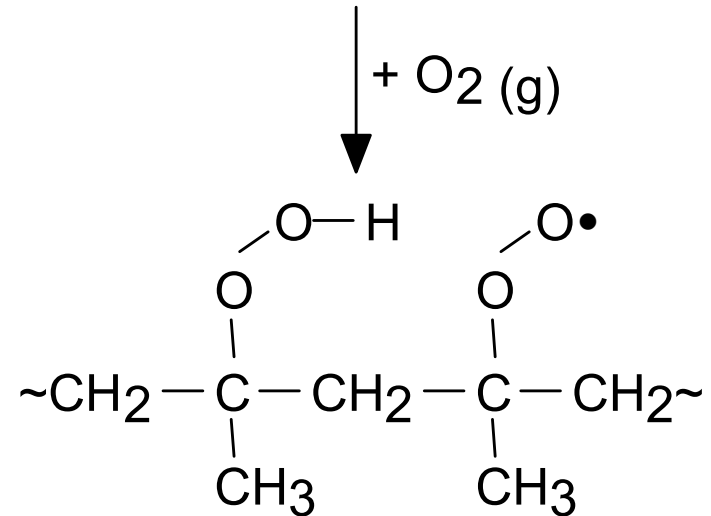
- **PROPAGATION**: Abundant  $\text{O}_2$  reacts with alkyl groups to produce “stable” peroxy radicals.  $\text{O}_3$  and  $\text{O}$  react to form unstable alkoxy radicals.

# PP SURFACE REACTIONS: PROPAGATION / AGING

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- **PROPAGATION / AGING**: Peroxy radicals abstract H from the PP chain, resulting in hydroperoxide, processes which take seconds to 10s minutes.



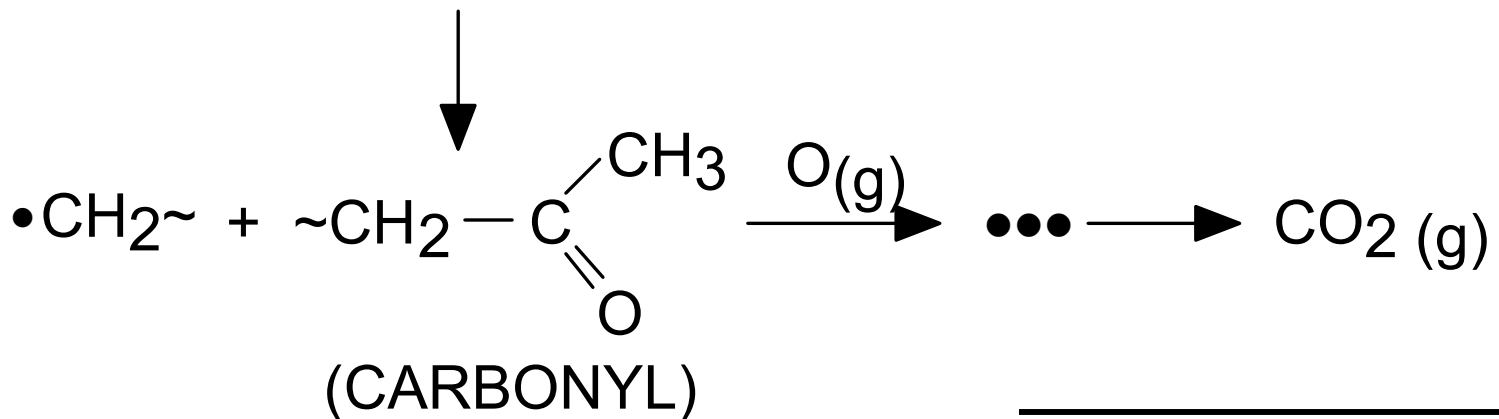
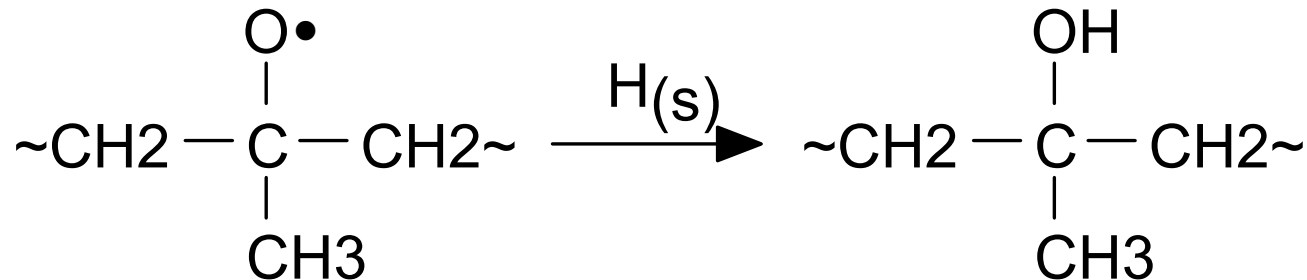
# PP SURFACE REACTION MECHANISM: TERMINATION

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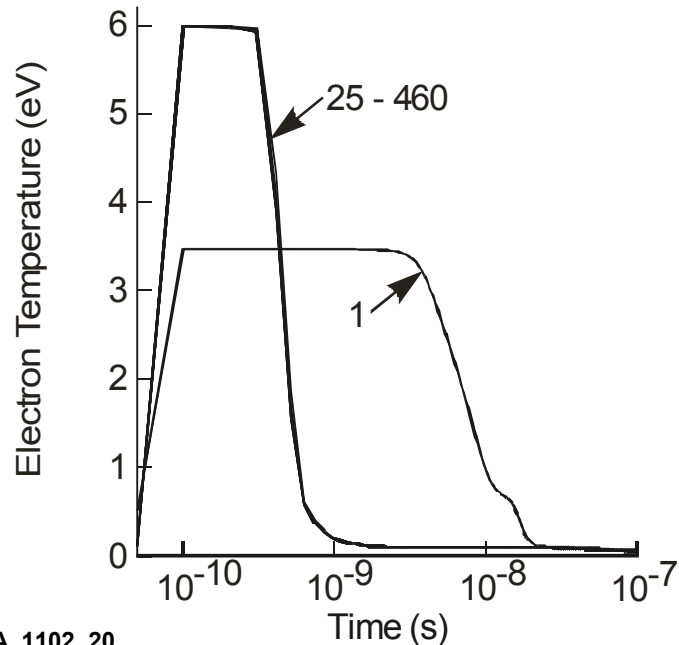
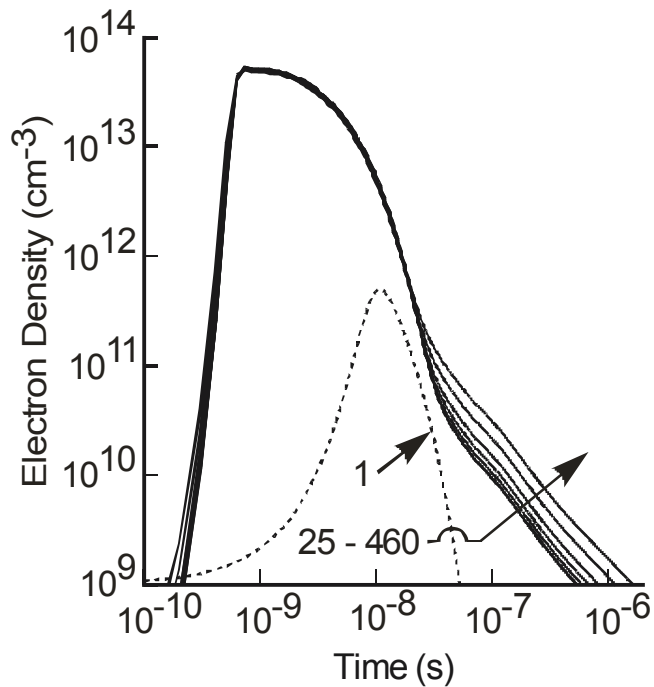
- **TERMINATION:** Alkoxy radicals react with the PP backbone to produce alcohols and carbonyls. Further reactions with O eventually erodes the film.

(ALKOXY RADICAL)

(ALCOHOLS)



# BASE CASE: $n_e$ , $T_e$

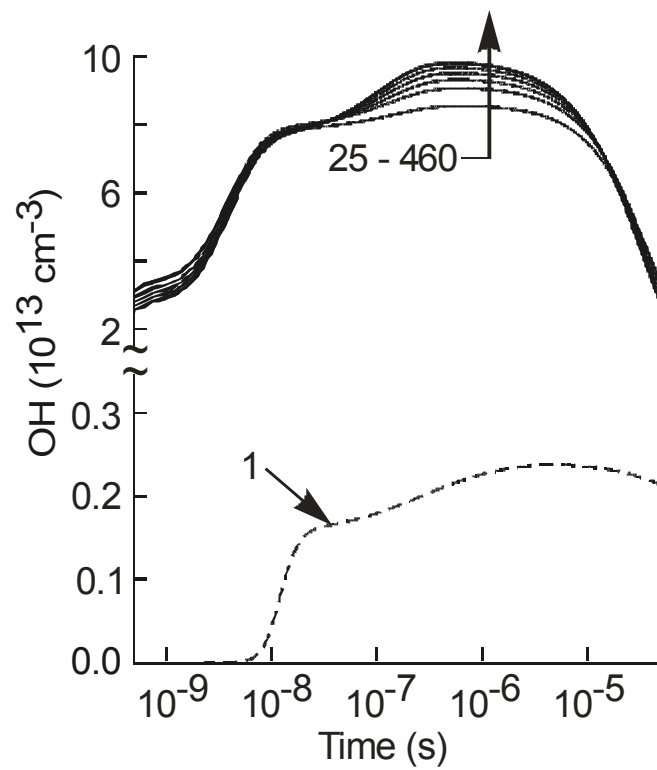
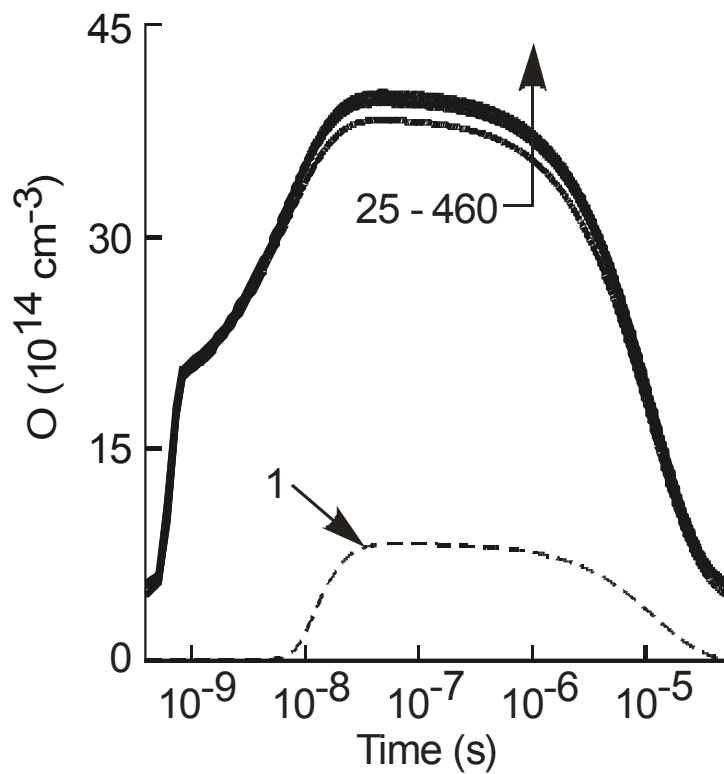


- Ionization is dominantly of  $N_2$  and  $O_2$ ,
  - $e + N_2 \rightarrow N_2^+ + e + e$ ,
  - $e + O_2 \rightarrow O_2^+ + e + e$ .
- After a few ns current pulse, electrons decay by attachment (primarily to  $O_2$ ).
- Dynamics of charging of the dielectrics produce later pulses with effectively larger voltages.
- $N_2/O_2/H_2O = 79/20/1$ , 300 K
- 15 kV, 9.6 kHz,  $0.8 \text{ J-cm}^{-2}$
- Web speed = 250 cm/s (460 pulses)



# GAS-PHASE RADICALS: O, OH

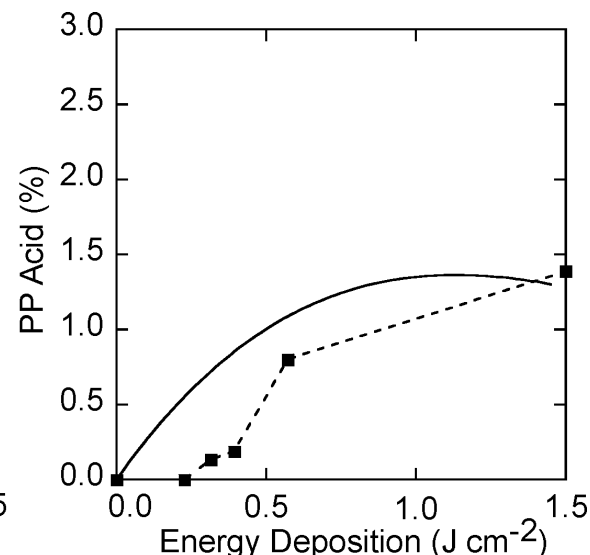
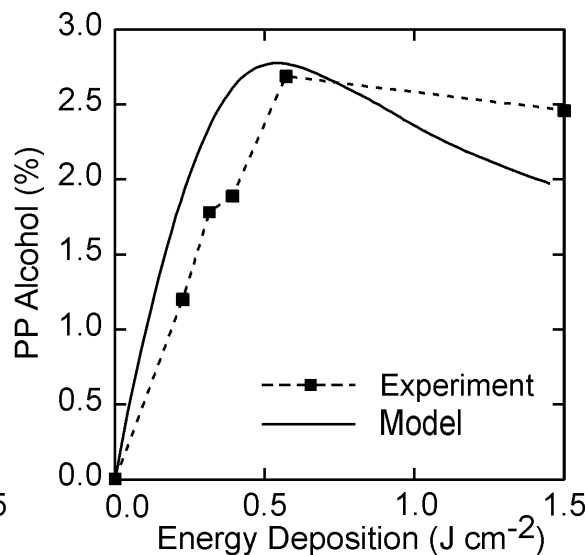
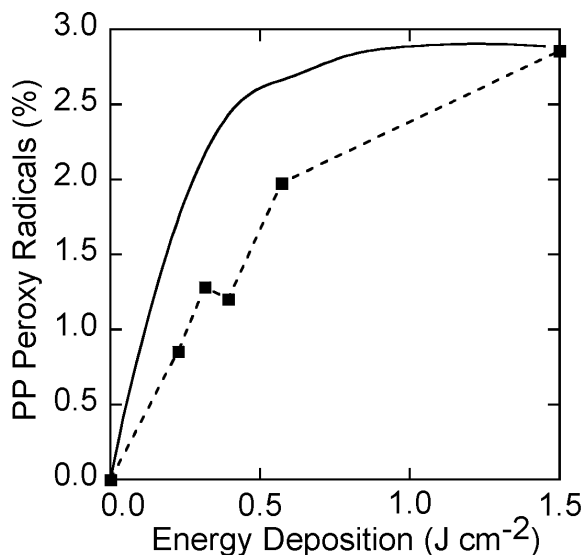
- Electron impact dissociation of  $O_2$  and  $H_2O$  produces O and OH. O is consumed in the primarily to form  $O_3$ ,
- After 100s of pulses, radicals attain a periodic steady state.



# PP SURFACE GROUPS vs ENERGY DEPOSITION

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- Surface concentrations of alcohols, peroxy radicals are near steady state with a few J-cm<sup>-2</sup>.
- Alcohol densities decrease at higher J-cm<sup>-2</sup> energy due to decomposition by O and OH to regenerate alkoxy radicals.



• Air, 300 K, 1 atm, 30% RH

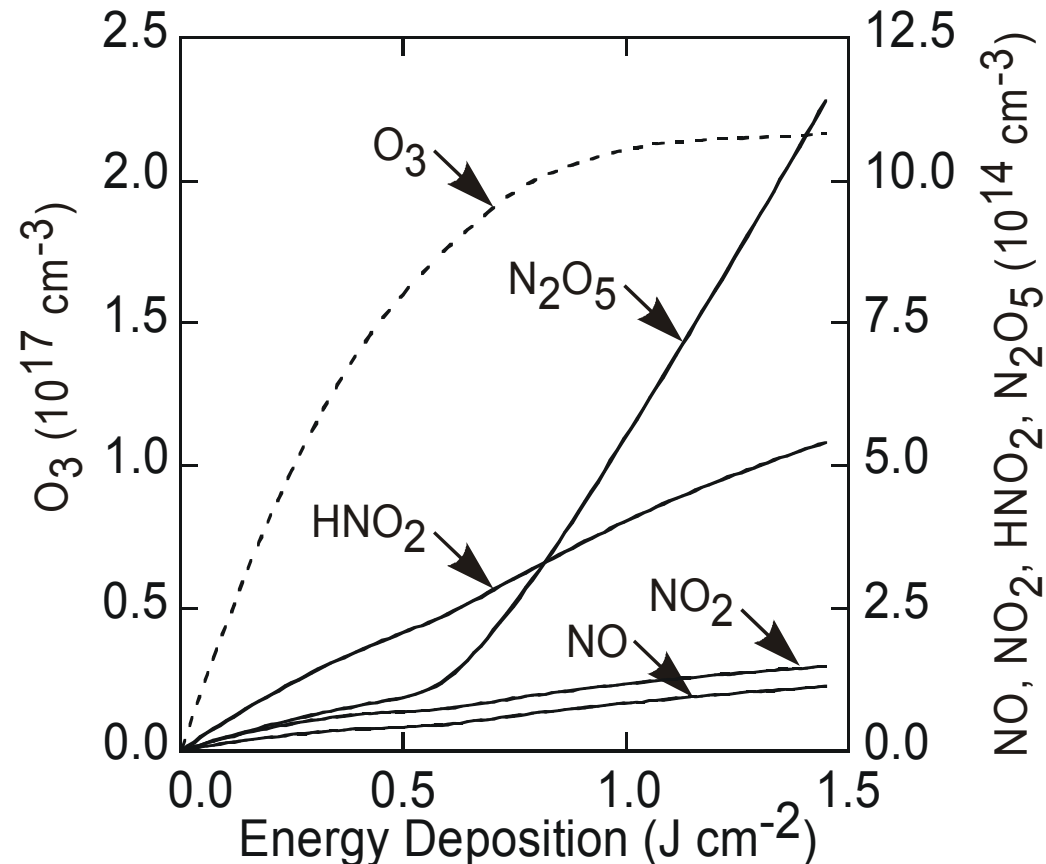
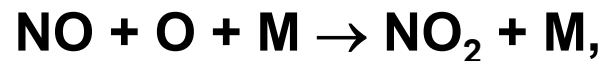
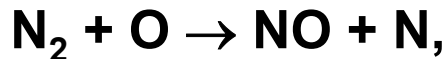
- Ref: L-A. Ohare *et al.*,  
Surf. Interface Anal. 33, 335 (2002).

# GAS-PHASE PRODUCTS: O<sub>3</sub>, N<sub>x</sub>O<sub>y</sub>, HNO<sub>x</sub>

- O<sub>3</sub> is produced by the reaction of O with O<sub>2</sub>,

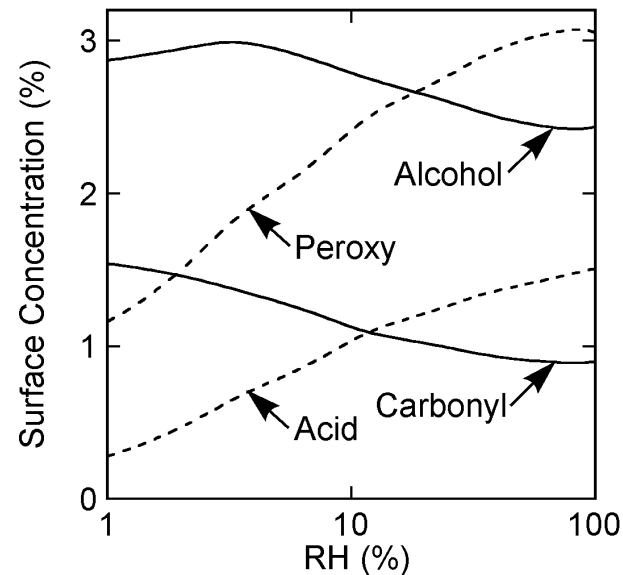
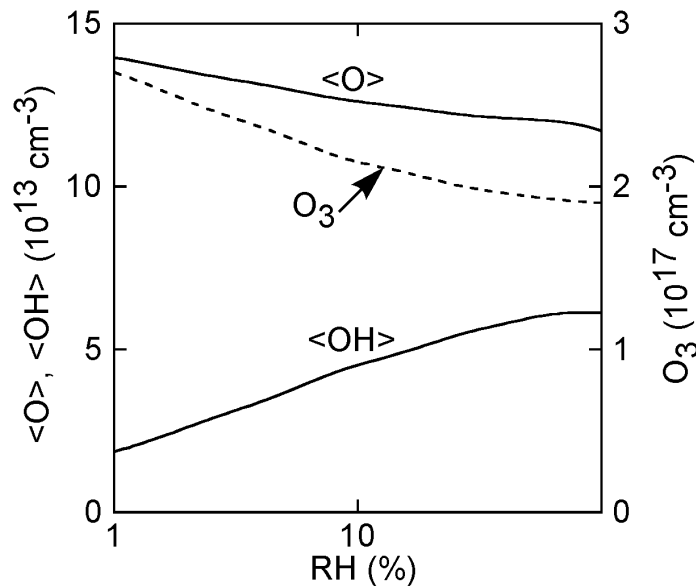
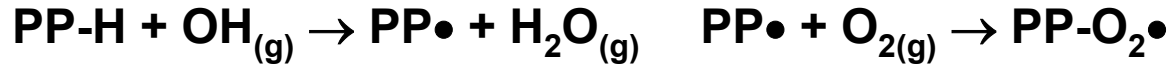


- N containing products include NO, NO<sub>2</sub>, HNO<sub>2</sub> and N<sub>2</sub>O<sub>5</sub>,

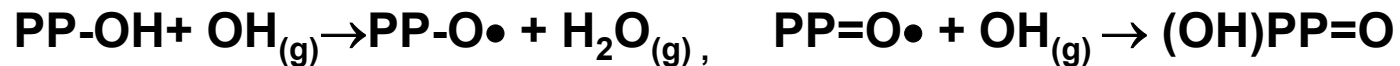


# HUMIDITY: PP FUNCTIONALIZATION BY OH

- Increasing RH produces more OH. Reactions with PP generate more alkyl radicals, rapidly converted to peroxy radicals by O<sub>2</sub>.

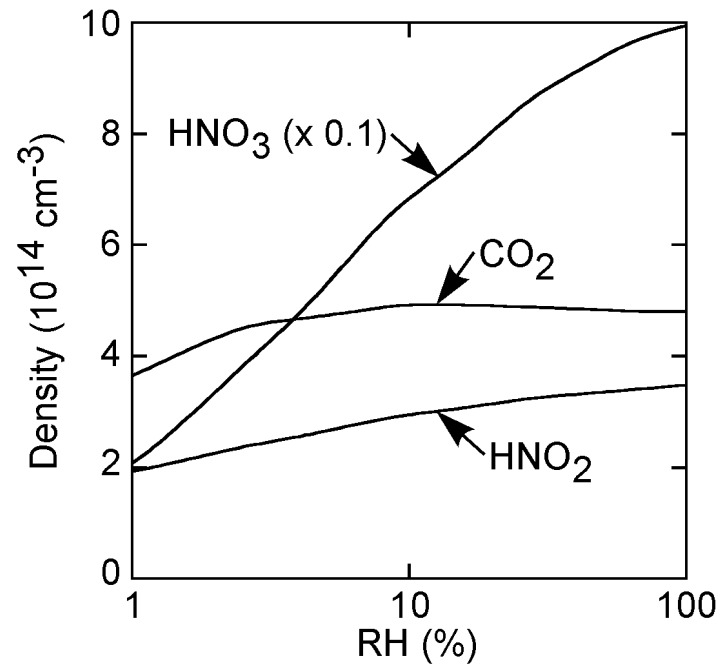
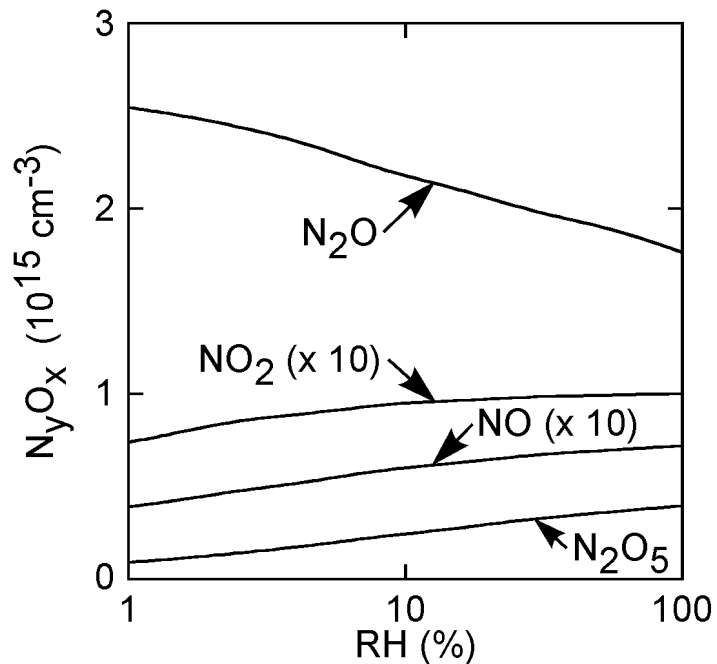
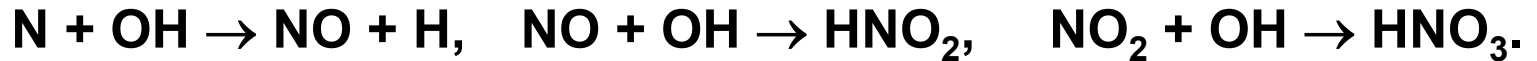


- Alcohol and carbonyl densities decrease due to increased consumption by OH to form alkoxy radicals and acids.



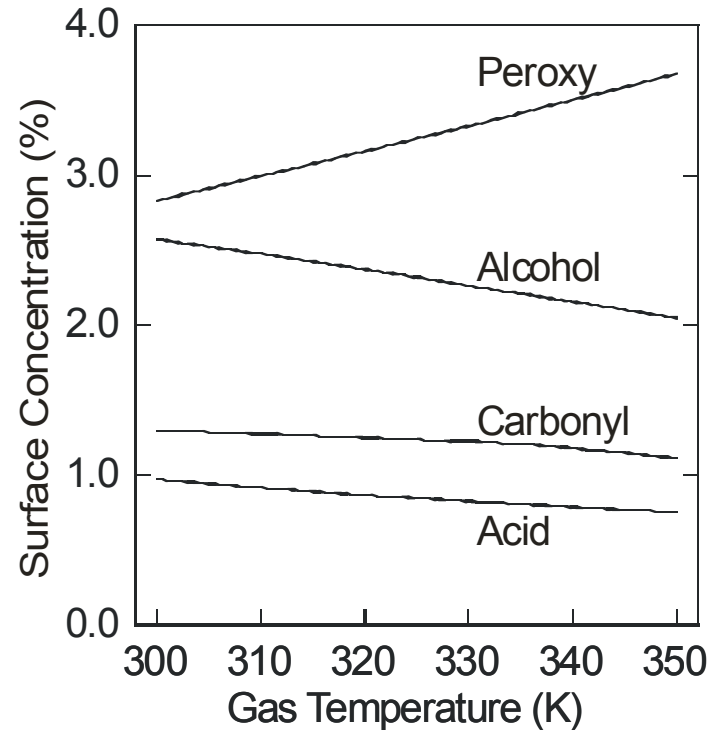
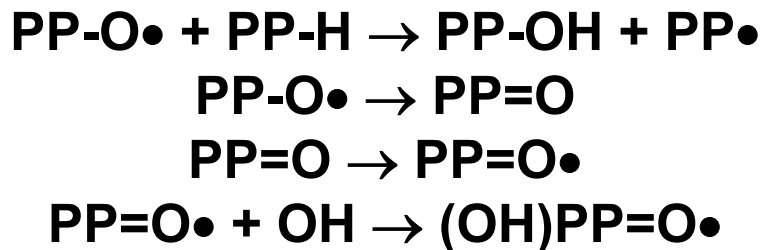
# EFFECT OF RH: GAS-PHASE PRODUCTS

- Higher RH results in decreasing O and increasing OH.
- Production of O<sub>3</sub> decreases while larger densities of HNO<sub>x</sub> are generated.



# EFFECT OF $T_{\text{Gas}}$ : PP FUNCTIONALIZATION

- Increasing  $T_{\text{gas}}$  decreases  $\text{O}_3$  leading to lower alkoxy production.
- $\text{PP}\bullet + \text{O}_{3(\text{g})} \rightarrow \text{PP-O}\bullet + \text{O}_{2(\text{g})}$ .
- ... and lower production of alcohols, carbonyl, and acids.

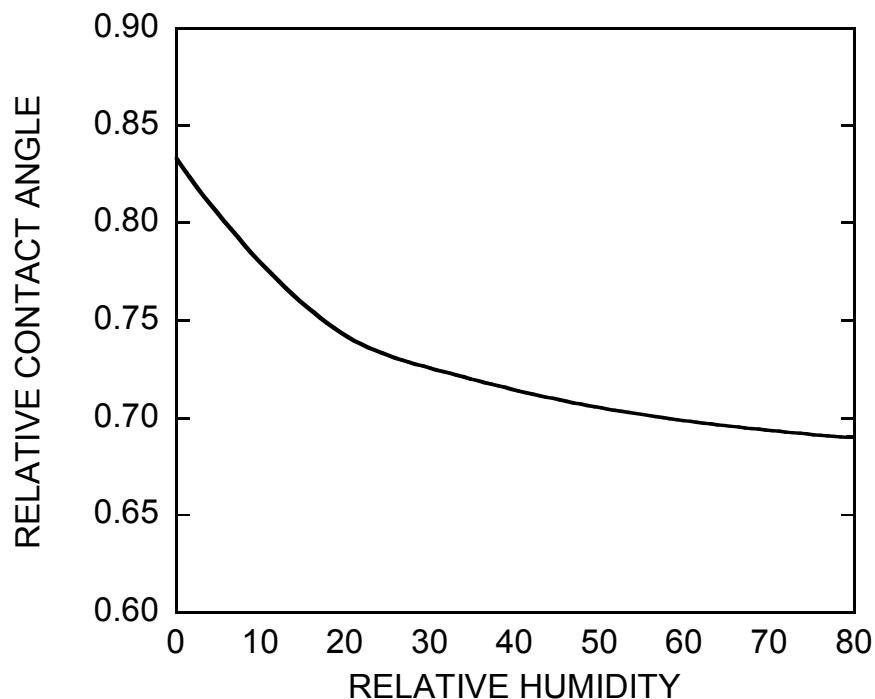


- Lower consumption of alkyl radicals by  $\text{O}_3$  enables reactions with  $\text{O}_2$  to dominate, increasing densities of peroxy.

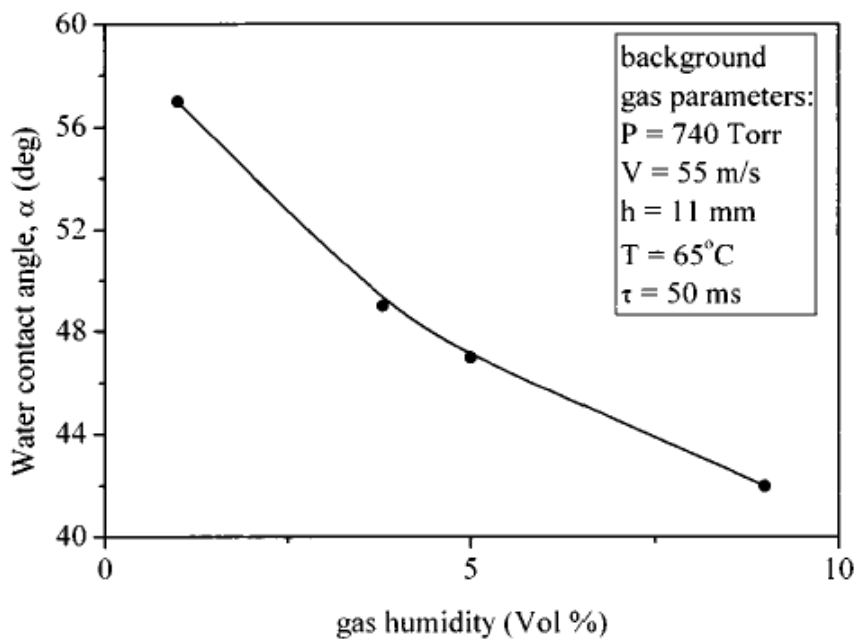
# RH: PREDICTED CONTACT ANGLE

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- Relation for wettability / contact angle vs concentration of functional groups is non-linear and poorly known.
- Assume wettability is mainly due to O on PP.



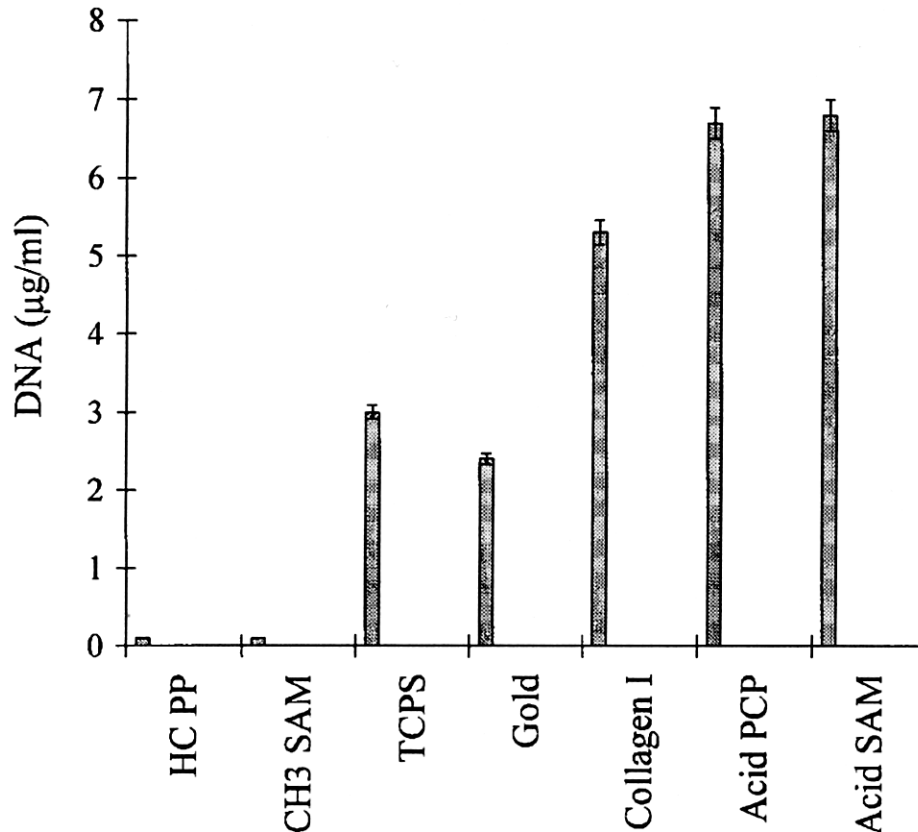
- **Model PP, Humid-air**



- **Polyethylene, Humid-air**
- Akishev, Plasmas Polym. 7, 261 (2002).

# WHAT'S THE UPSIDE: BETTER FRITO BAGS OR ENGINEERED BIOCOMPATIBLE COATINGS?

- *The ability to control functional groups on polymers through fundamental understanding of plasma-solid interactions opens the realm of engineered large area specialty surfaces.*



- Keratinocyte cells adhere to hydrocarbon polymers containing carboxylic acid groups (PCP, SAM).

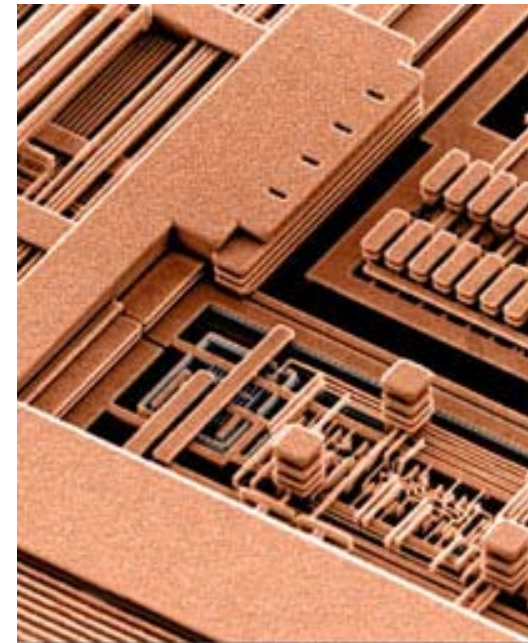
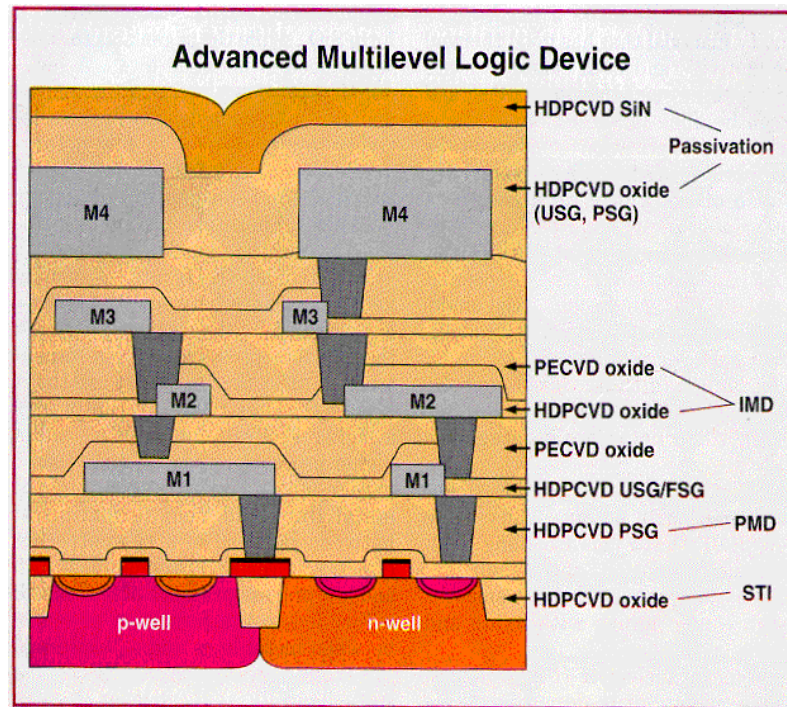
- Haddow et al., J. Biomed. Mat. Res. 47, 379 (1999)



# PLASMAS AND MICROELECTRONICS FABRICATION

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- The striking improvement in the functionality of microelectronics devices results from shrinking of individual components and increasing complexity of the circuitry

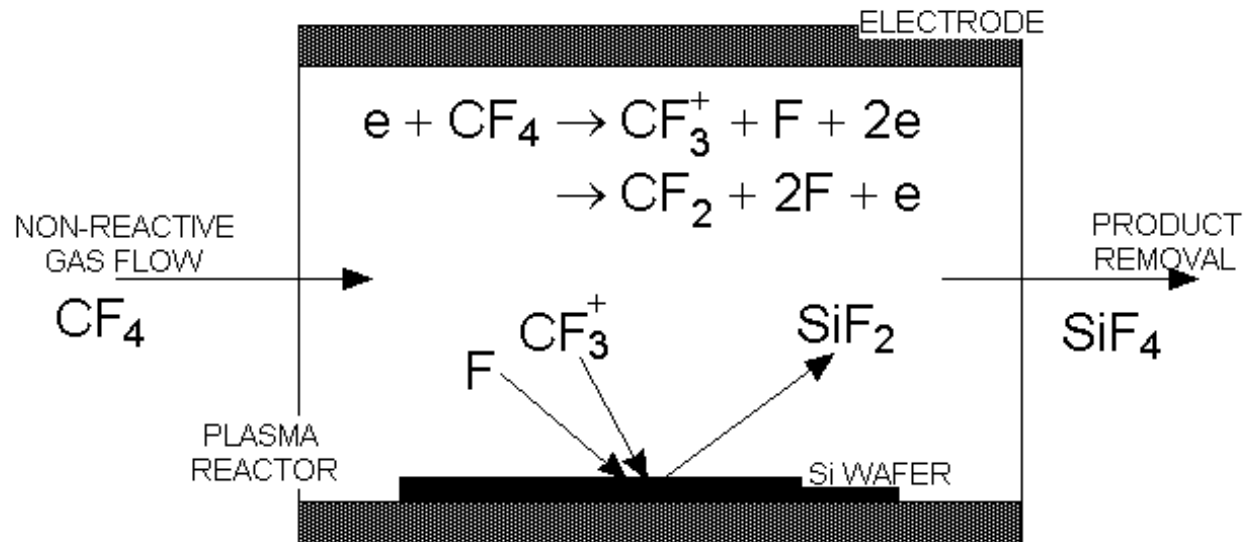


Ref: IBM Microelectronics

- Plasmas are absolutely essential to the fabrication of microelectronics.

# PLASMAS IN MICROELECTRONICS FABRICATION

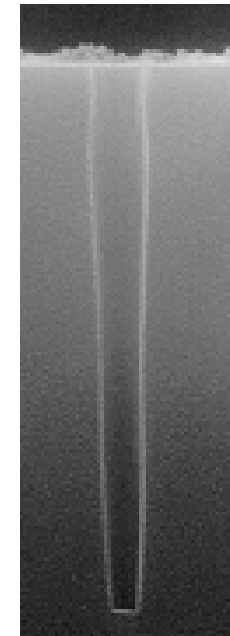
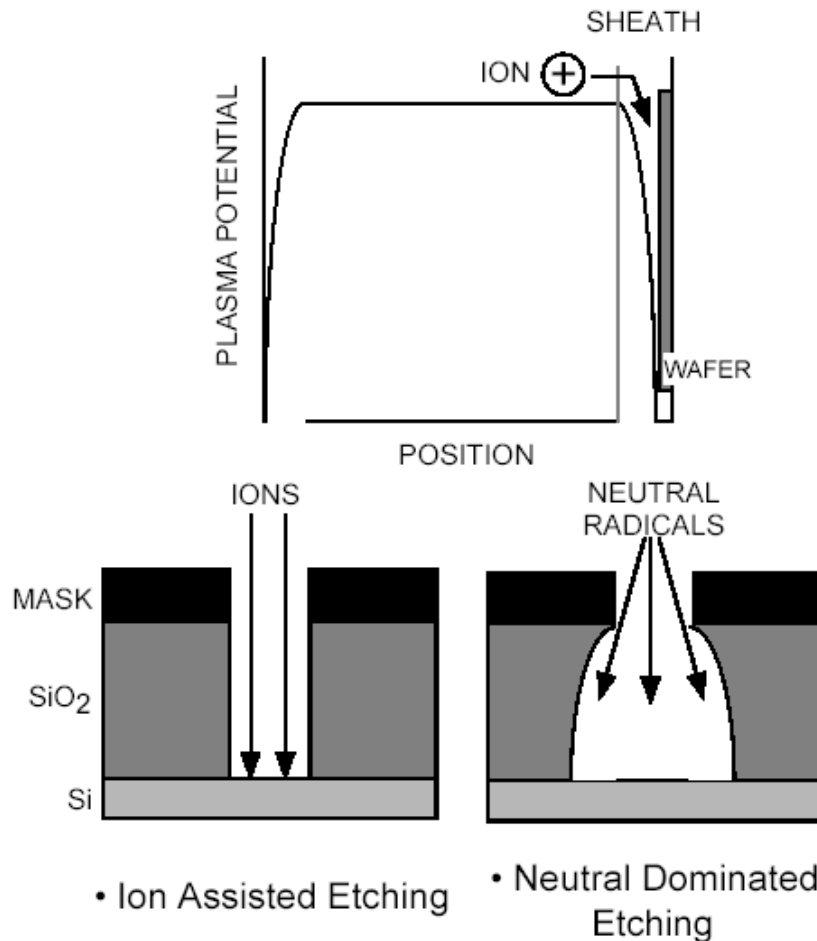
- Plasmas play a dual role in microelectronics fabrication.
- First, electron impact on otherwise unreactive gases produces neutral radicals and ions.



- These species then drift or diffuse to surfaces where they add, remove or modify materials.

# PLASMAS IN MICROELECTRONICS FABRICATION

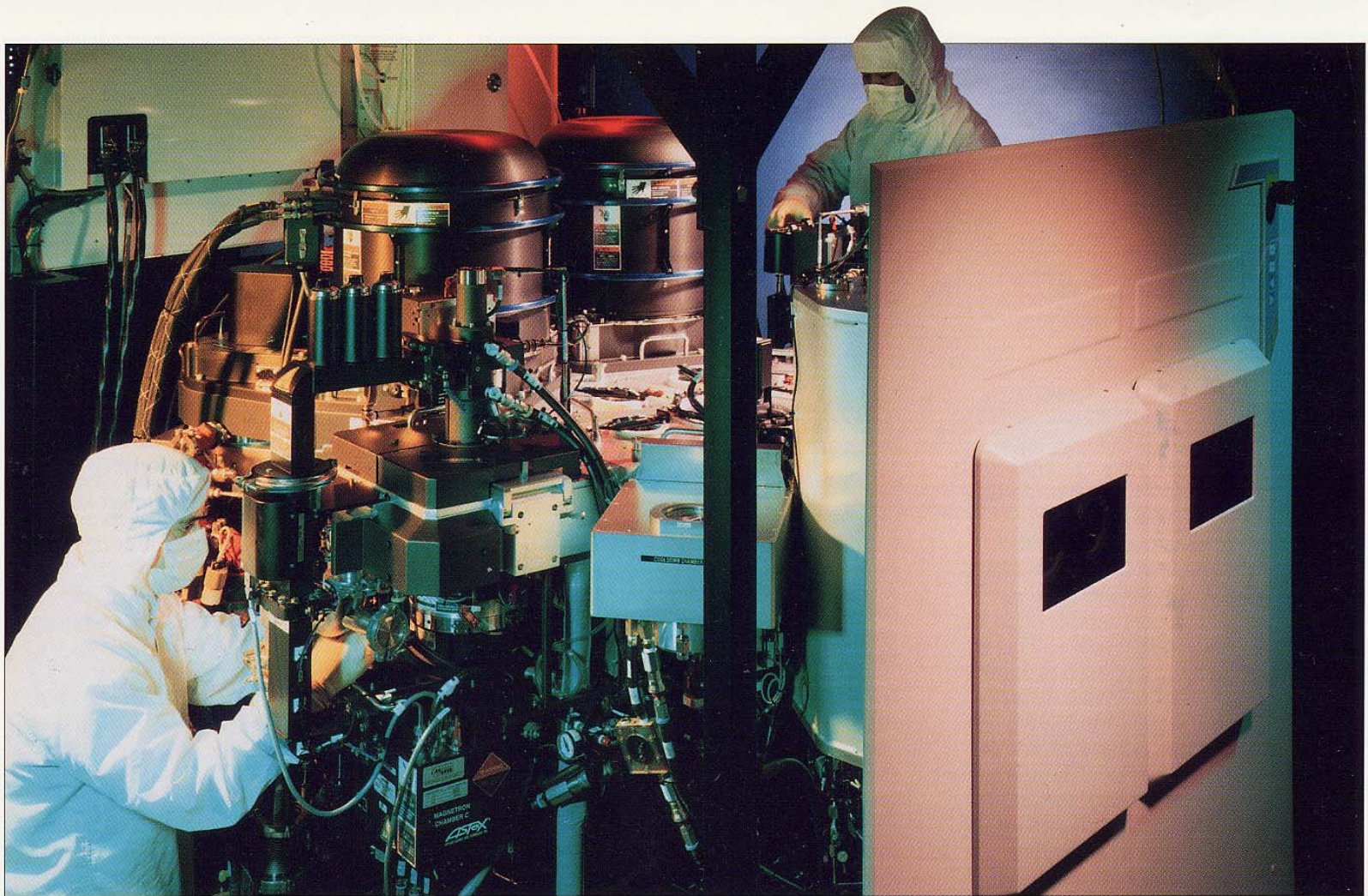
- Second, ions deliver directed activation energy to surfaces fabricating fine having extreme and reproducible tolerances.



- 0.25  $\mu\text{m}$  Feature (C. Cui, AMAT)

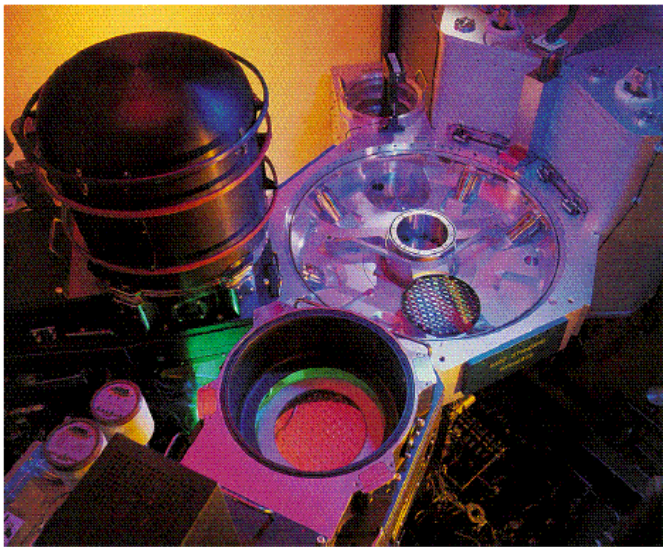
# APPLIED MATERIALS DECOUPLED PLASMA SOURCES (DPS)

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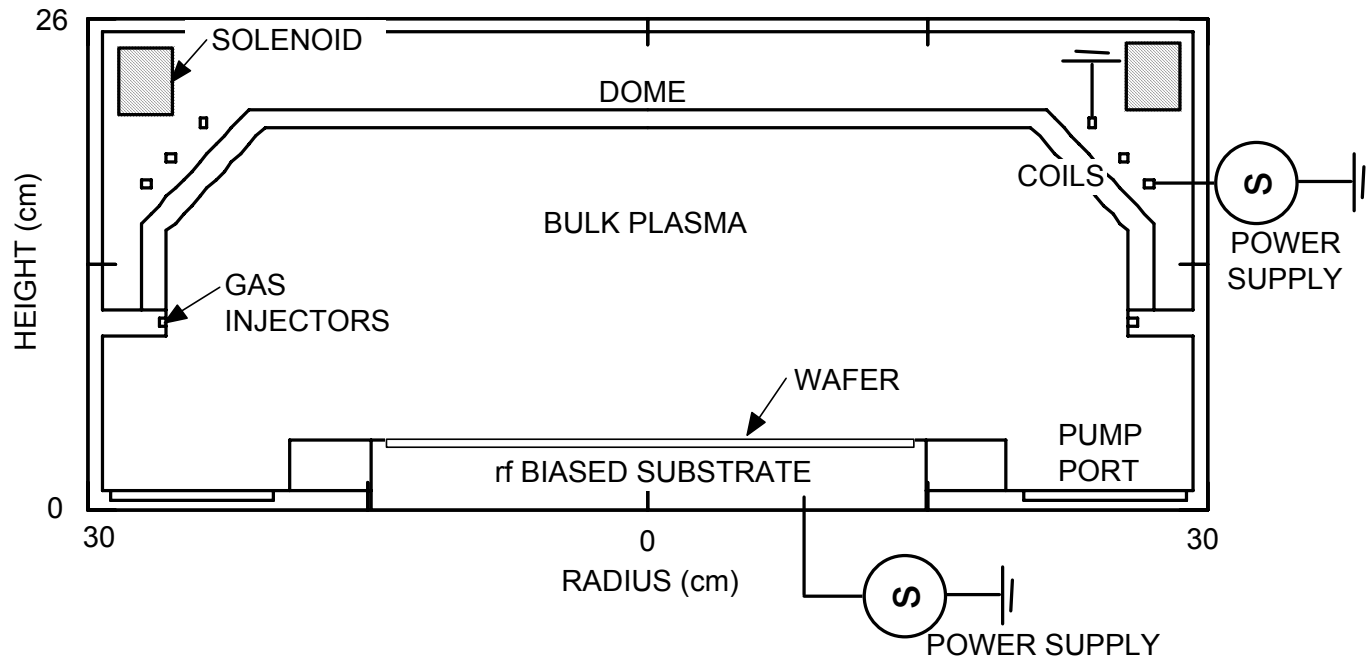
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Optical and Discharge Physics



# rf BIASED INDUCTIVELY COUPLED PLASMAS

- Inductively Coupled Plasmas (ICPs) with rf biasing are used here.
- < 10s mTorr, 10s MHz, 100s W – kW, electron densities of  $10^{11}$ - $10^{12}$  cm<sup>-3</sup>.

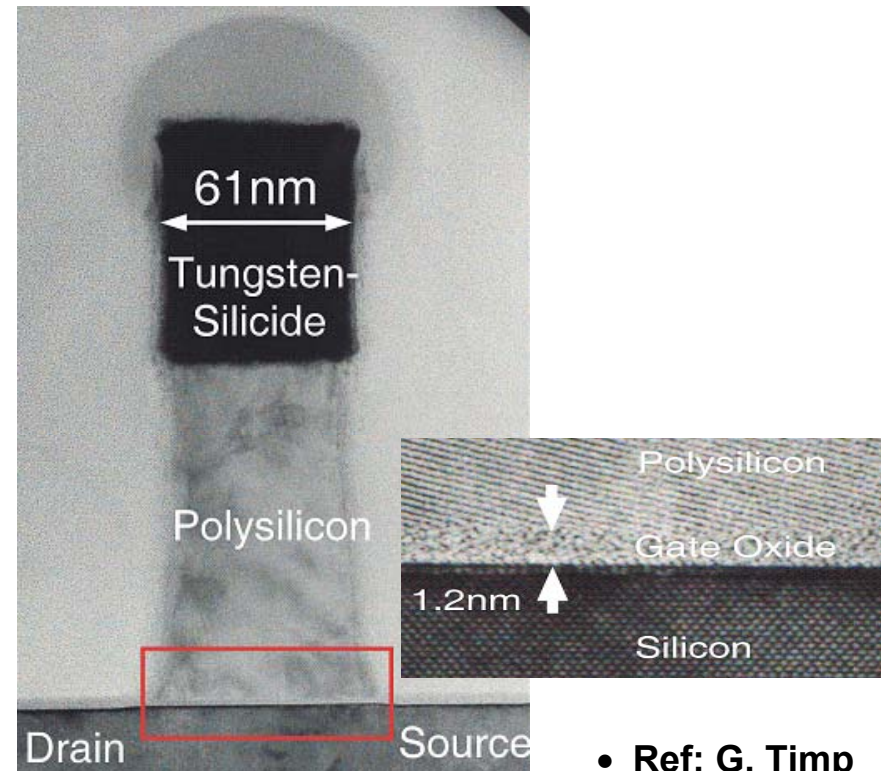


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# SELECTIVITY IN MICROELECTRONICS FABRICATION: PLASMAS AND POLYMERS

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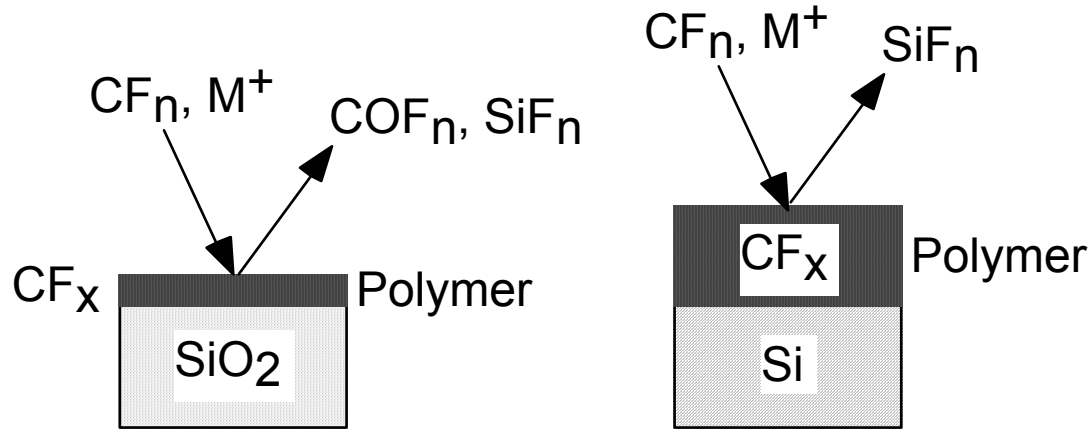
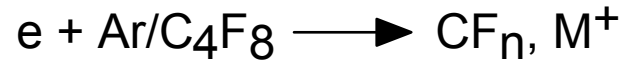
- Fabricating complex microelectronic structures made of different materials requires extreme *selectivity* in, for example, etching Si with respect to  $\text{SiO}_2$ .
- Monolayer selectivity is required in advanced etching processes.
- These goals are met by the unique plasma-polymer interactions enabled in fluorocarbon chemistries.



# FLUOROCARBON PLASMA ETCHING: SELECTIVITY

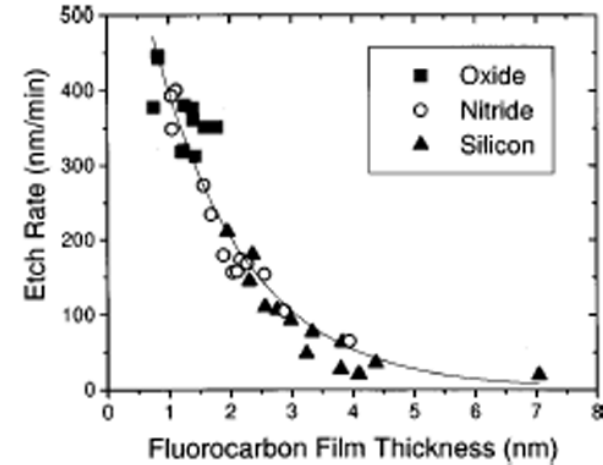
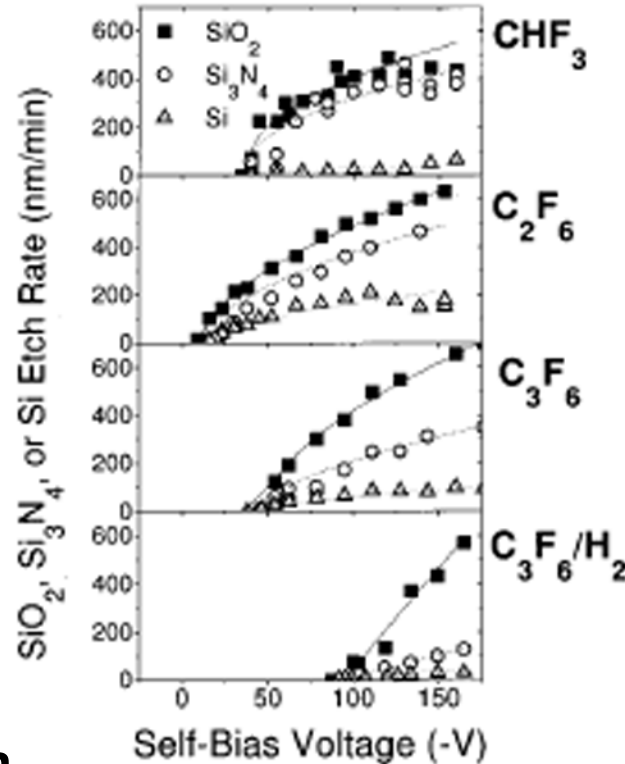
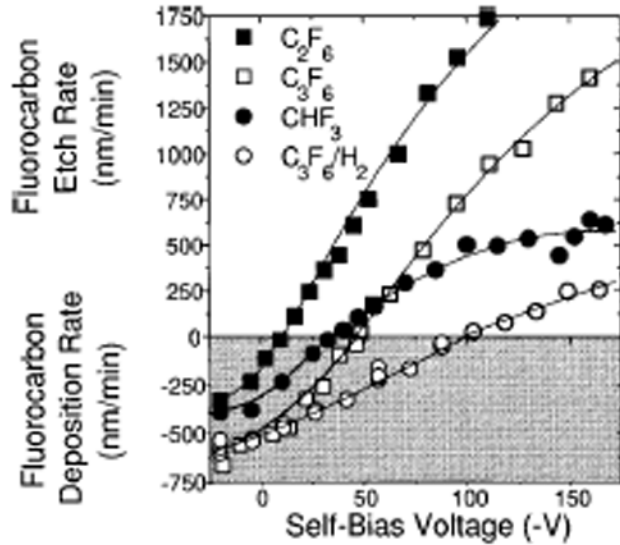
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- Selectivity in fluorocarbon etching relies on polymer deposition.
- Electron impact dissociation of feedstock fluorocarbons produce polymerizing radicals and ions, resulting in polymer deposition.



- Compound dielectrics contain oxidants which consume the polymer, producing thinner polymer layers.
- Thicker polymer on non-dielectrics restrict delivery of ion energy (lower etching rates).

# FLUOROCARBON PLASMA ETCHING: SELECTIVITY



- Low bias: Deposition
- High bias: etching
- Etch Rate ( $SiO_2 > Si$ )

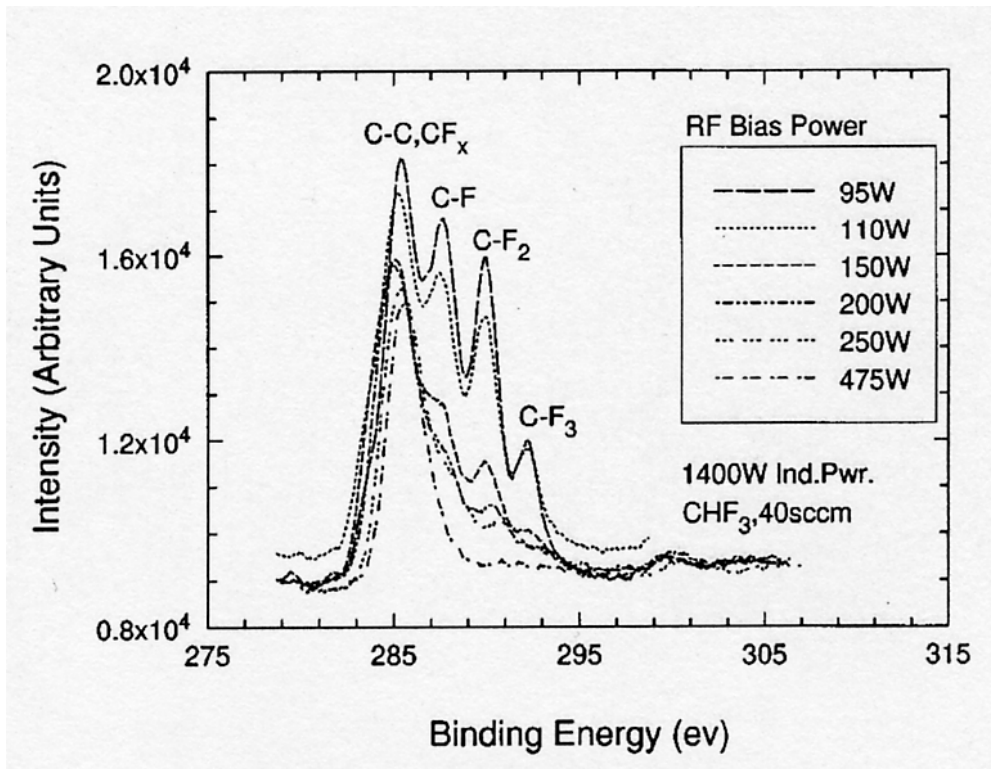
- Polymer Thickness ( $SiO_2 < Si$ )

• G. Oerhlein, et al., JVSTA 17, 26 (1999)



# FLUOROCARBON PLASMA ETCHING: POLYMER

- The polymer composition deposited in fluorocarbon plasmas depends on feedstock, pressure, power, bias power.

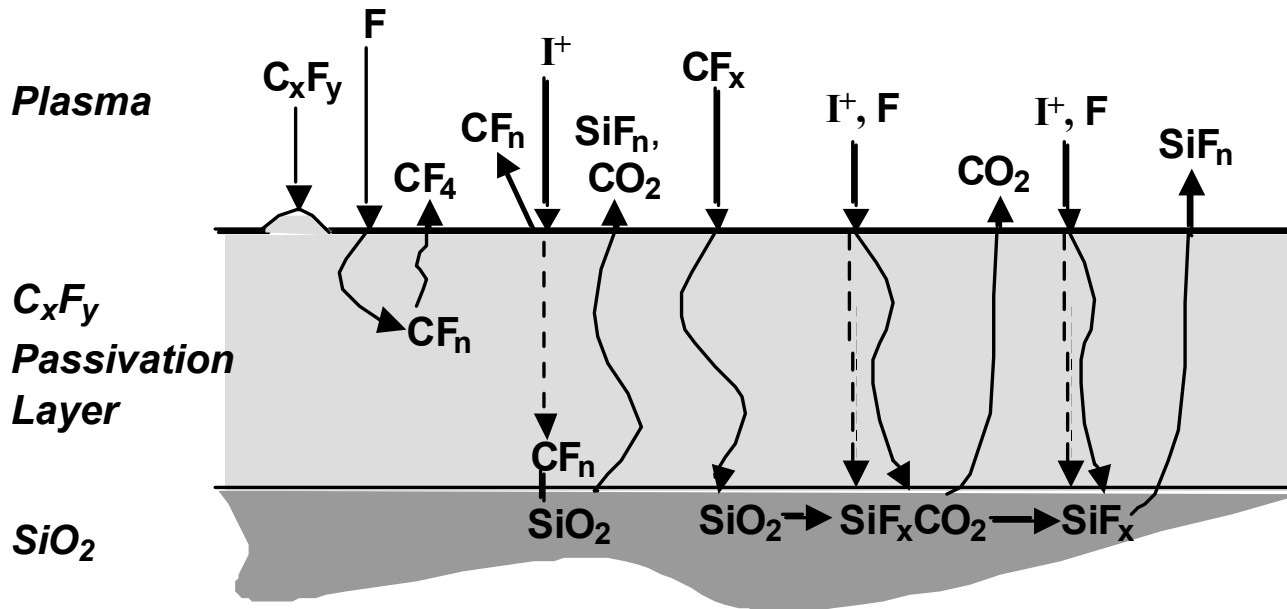


- Rueger et al., JVST A 15, 1881 (1997)

- For discussion PTFE (poly)tetrafluoroethylene [C<sub>2</sub>F<sub>4</sub>]<sub>n</sub> is a good approximation for most layers.

# SURFACE KINETICS: FLUOROCARBON PLASMA ETCHING Si/SiO<sub>2</sub>

- C<sub>x</sub>F<sub>y</sub> passivation regulates delivery of precursors and activation energy.
- Chemisorption of CF<sub>x</sub> produces a complex at the oxide-polymer interface.
- 2-step ion activated (through polymer layer) etching of the complex consumes the polymer. Activation scales inversely with polymer thickness.
- Etch precursors and products diffuse through the polymer layer.



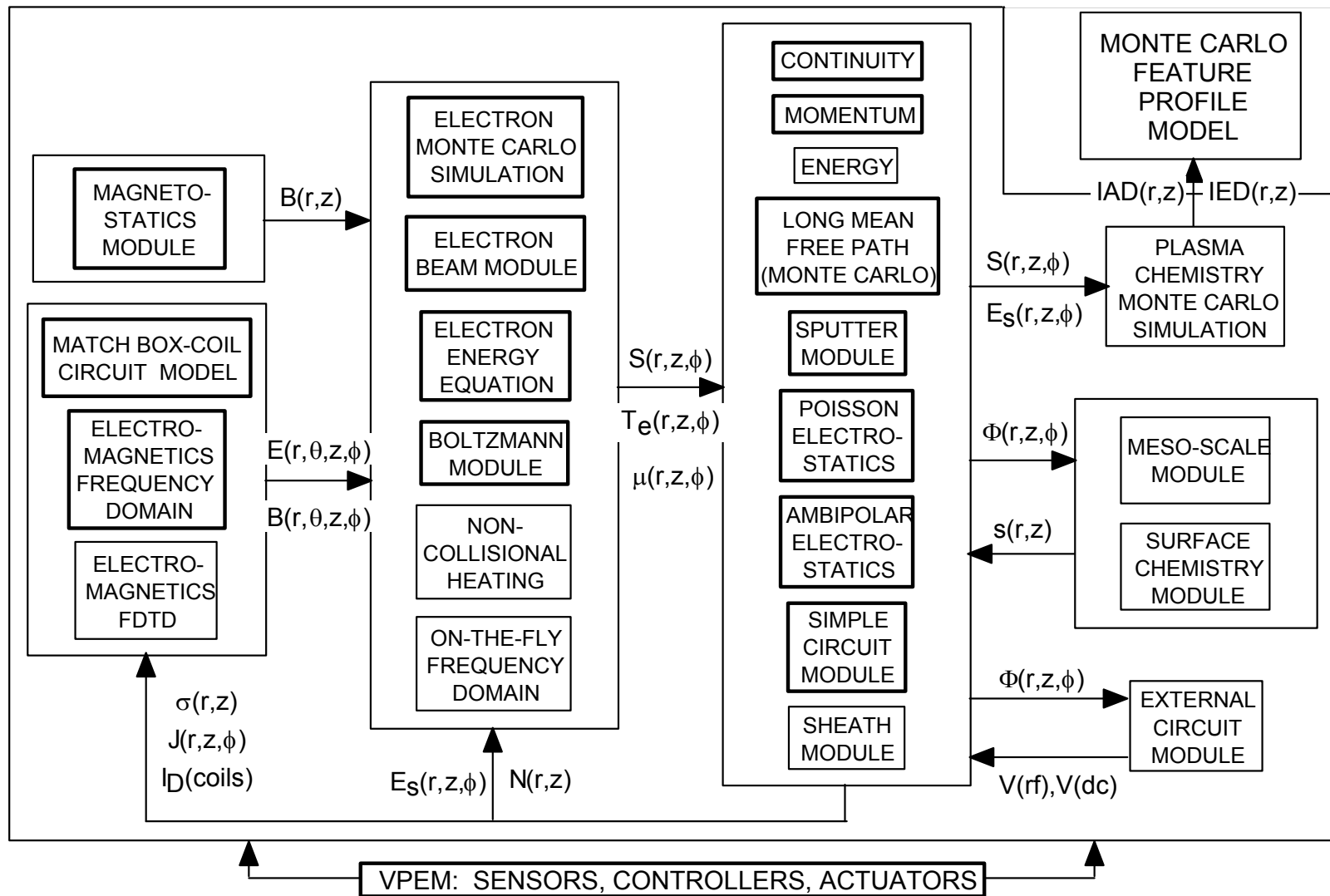
- In Si etching, CF<sub>x</sub> is not consumed, resulting in thicker polymer layers.

# MODELING OF FLUOROCARBON PLASMA ETCHING

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- Our research group has developed an integrated reactor and feature scale modeling hierarchy to model plasma processing systems.
- HPEM (Hybrid Plasma Equipment Model)
  - Reactor scale
  - 2- and 3-dimensional
  - ICP, CCP, MERIE, ECR
  - Surface chemistry
  - First principles
- MCFPM (Monte Carlo Feature Profile Model)
  - Feature scale
  - 2- and 3-dimensional
  - Fluxes from HPEM
  - First principles

# HYBRID PLASMA EQUIPMENT MODEL



# ELECTROMAGNETICS MODULE

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- The wave equation is solved in the frequency domain using sparse matrix techniques (2D,3D):

$$-\nabla \cdot \left( \frac{1}{\mu} \nabla \cdot \bar{E} \right) + \nabla \cdot \left( \frac{1}{\mu} \nabla \bar{E} \right) = \frac{\partial^2 (\epsilon \bar{E})}{\partial t^2} + \frac{\partial (\bar{\sigma} \cdot \bar{E} + \bar{J})}{\partial t}$$

$$\vec{E}(\vec{r}, t) = \vec{E}'(\vec{r}) \exp(-i(\omega t + \varphi(\vec{r})))$$

- Conductivities are tensor quantities (2D,3D):

$$\bar{\sigma} = \sigma_o \frac{m v_m}{q \alpha} \frac{1}{\left( \alpha^2 + |\vec{B}|^2 \right)} \begin{pmatrix} \alpha^2 + B_r^2 & \alpha B_z + B_r B_\theta & -\alpha B_\theta + B_r B_z \\ -\alpha B_z + B_r B_\theta & \alpha^2 + B_\theta^2 & \alpha B_r + B_\theta B_z \\ -\alpha B_\theta + B_r B_z & -\alpha B_r + B_\theta B_z & \alpha^2 + B_z^2 \end{pmatrix}$$

$$\vec{j} = \bar{\sigma} \cdot \vec{E} \quad \alpha = \frac{(i\omega + v_m)}{q/m}, \quad \sigma_o = \frac{q^2 n_e}{m v_m}$$

# ELECTRON ENERGY TRANSPORT

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- Continuum (2D,3D):

$$\partial \left( \frac{3}{2} n_e k T_e \right) / \partial t = S(T_e) - L(T_e) - \nabla \cdot \left( \frac{5}{2} \Phi k T_e - \bar{\kappa}(T_e) \cdot \nabla T_e \right) + S_{EB}$$

where $S(T_e)$	=	Power deposition from electric fields
$L(T_e)$	=	Electron power loss due to collisions
$\Phi$	=	Electron flux
$\kappa(T_e)$	=	Electron thermal conductivity tensor
$S_{EB}$	=	Power source source from beam electrons

- Power deposition has contributions from wave and electrostatic heating.
- Kinetic (2D,3D): A Monte Carlo Simulation is used to derive  $f(\varepsilon, \vec{r}, t)$  including electron-electron collisions using electromagnetic fields from the EMM and electrostatic fields from the FKM.

# PLASMA CHEMISTRY, TRANSPORT AND ELECTROSTATICS

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- Continuity, momentum and energy equations are solved for each species (with jump conditions at boundaries) (2D,3D).

$$\frac{\partial N_i}{\partial t} = -\nabla \cdot (N_i \vec{v}_i) + S_i$$

$$\frac{\partial (N_i \vec{v}_i)}{\partial t} = \frac{1}{m_i} \nabla (k N_i T_i) - \nabla \cdot (N_i \vec{v}_i \vec{v}_i) + \frac{q_i N_i}{m_i} (\vec{E} + \vec{v}_i \times \vec{B}) - \nabla \cdot \bar{\mu}_i - \sum_j \frac{m_j}{m_i + m_j} N_i N_j (\vec{v}_i - \vec{v}_j) \nu_{ij}$$

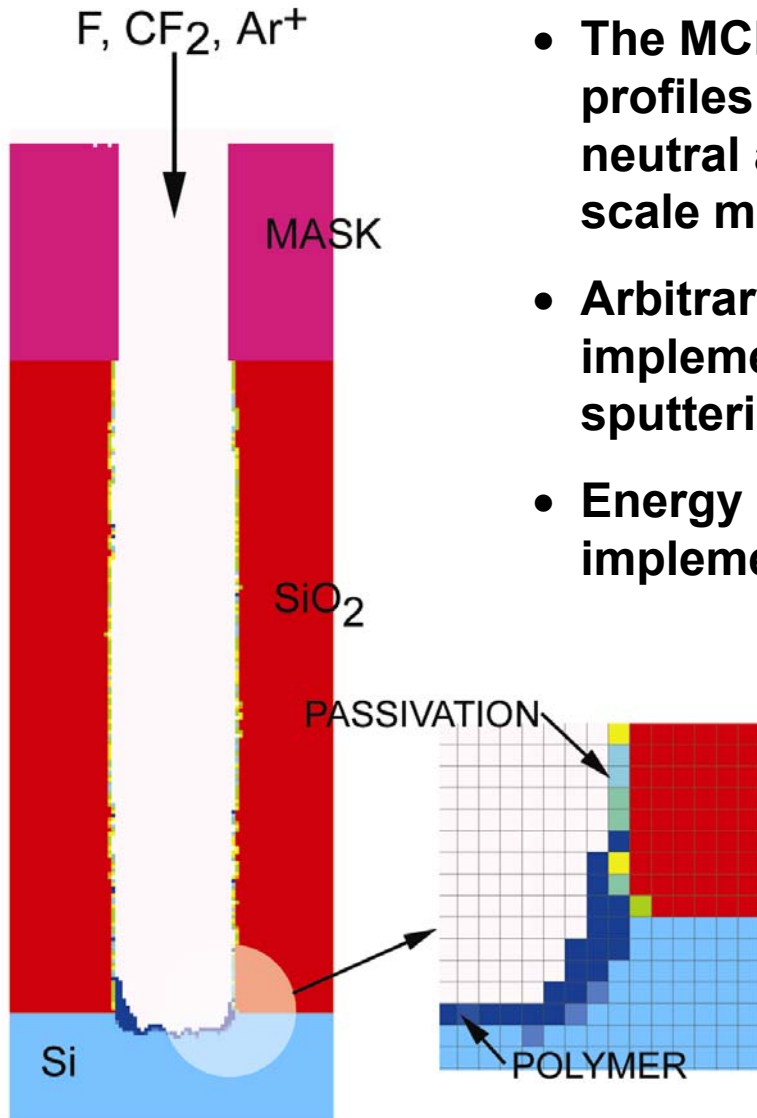
$$\begin{aligned} \frac{\partial (N_i \varepsilon_i)}{\partial t} + \nabla \cdot Q_i + P_i \nabla \cdot U_i + \nabla \cdot (N_i U_i \varepsilon_i) &= \frac{N_i q_i^2 \nu_i}{m_i (\nu_i^2 + \omega^2)} E^2 \\ &+ \frac{N_i q_i^2}{m_i \nu_i} E_s^2 + \sum_j 3 \frac{m_{ij}}{m_i + m_j} N_i N_j R_{ij} k_B (T_j - T_i) \pm \sum_j 3 N_i N_j R_{ij} k_B T_j \end{aligned}$$

- Implicit solution of Poisson's equation (2D,3D):

$$\nabla \cdot \varepsilon \nabla \Phi(t + \Delta t) = - \left( \rho_s + \sum_i q_i N_i - \Delta t \cdot \sum_i (q_i \nabla \cdot \vec{\phi}_i) \right)$$

# MONTE CARLO FEATURE PROFILE MODEL (MCFPM)

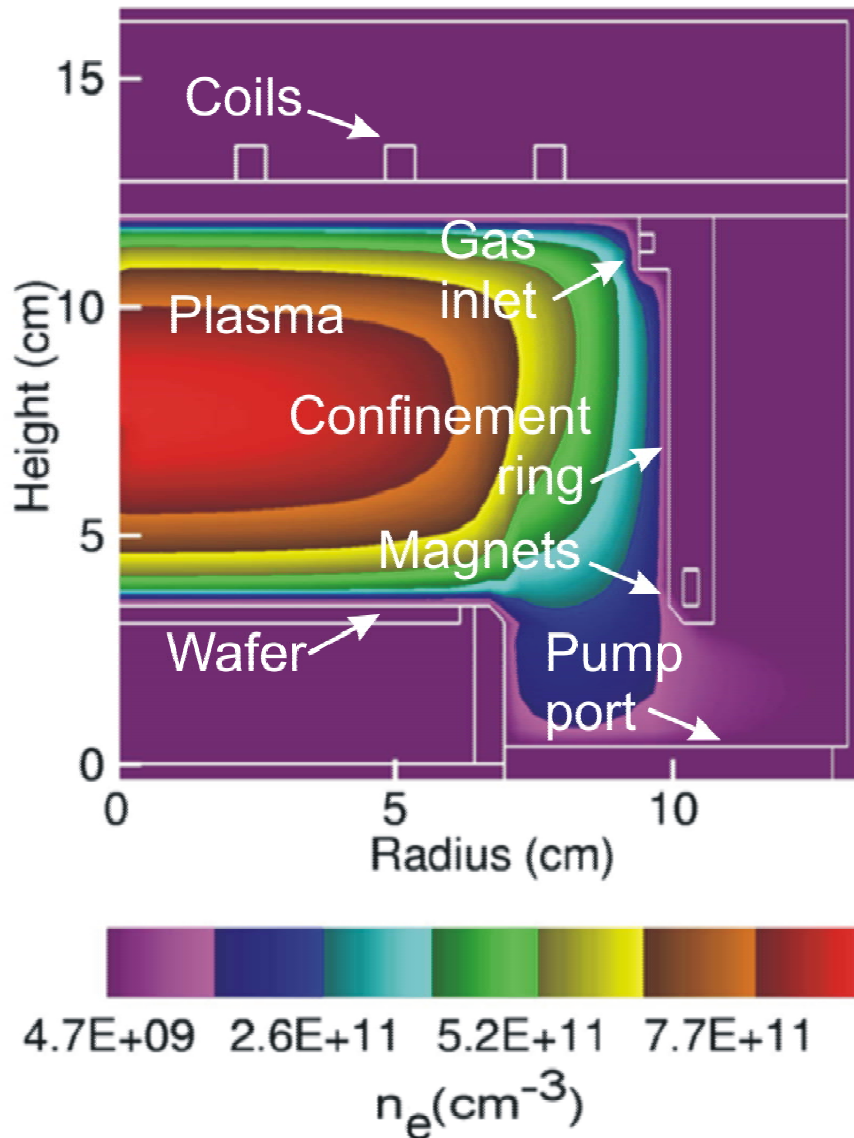
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- The MCFPM predicts time and spatially dependent profiles using energy and angularly resolved neutral and ion fluxes obtained from equipment scale models.
- Arbitrary chemical reaction mechanisms may be implemented, including thermal and ion assisted, sputtering, deposition and surface diffusion.
- Energy and angular dependent processes are implemented using parametric forms.
- Mesh centered identify of materials allows “burial”, overlayers and transmission of energy through materials.

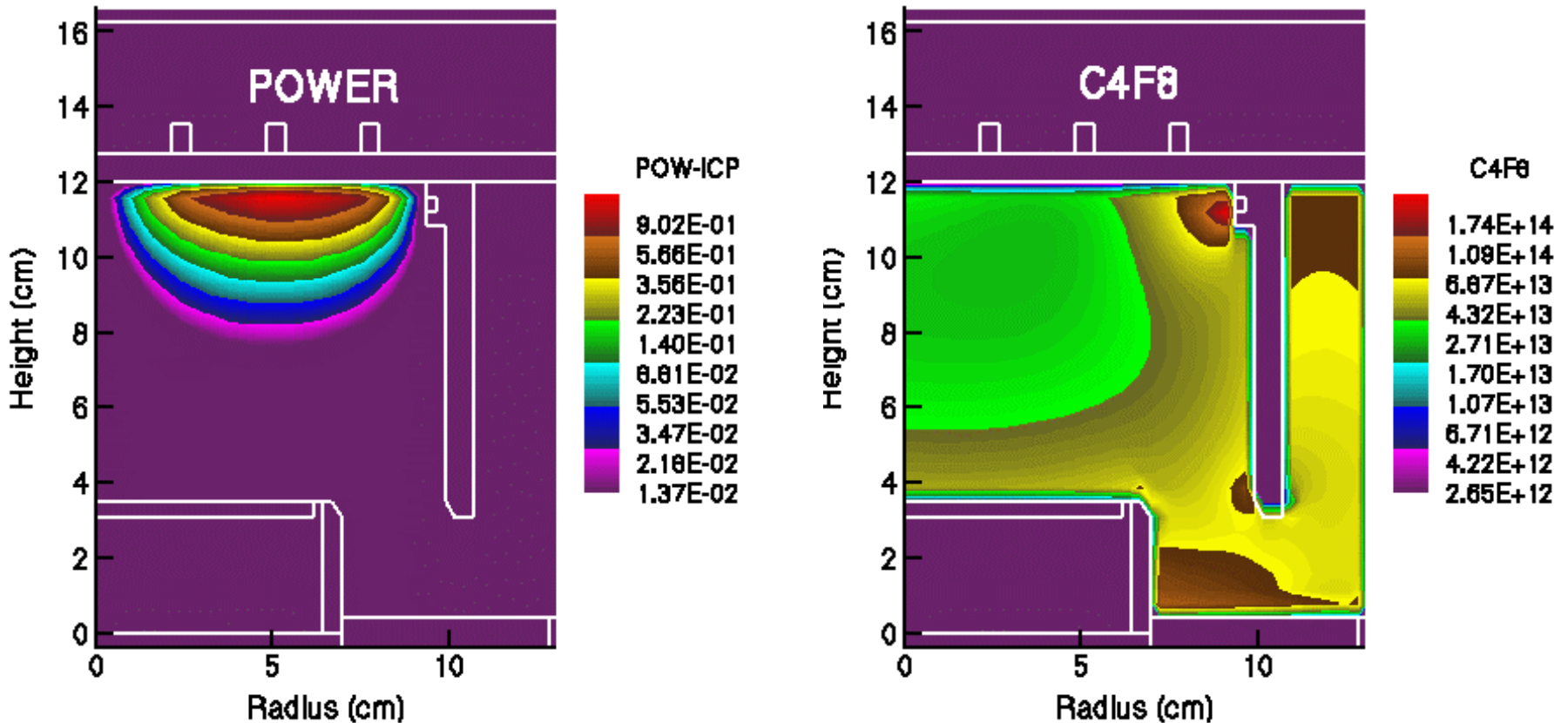


# TYPICAL ICP CONDITIONS: [e] FOR C<sub>4</sub>F<sub>8</sub>, 10 mTORR



- An ICP reactor patterned after Oeherlein, et al. was used for validation.
- Reactor uses 3-turn coil (13.56 MHz) with rf biased substrate (3 MHz)
- Electron densities are  $10^{11}$ - $10^{12}$  cm<sup>-3</sup> for 1.4 kW.

# POWER, C<sub>4</sub>F<sub>8</sub> DENSITY

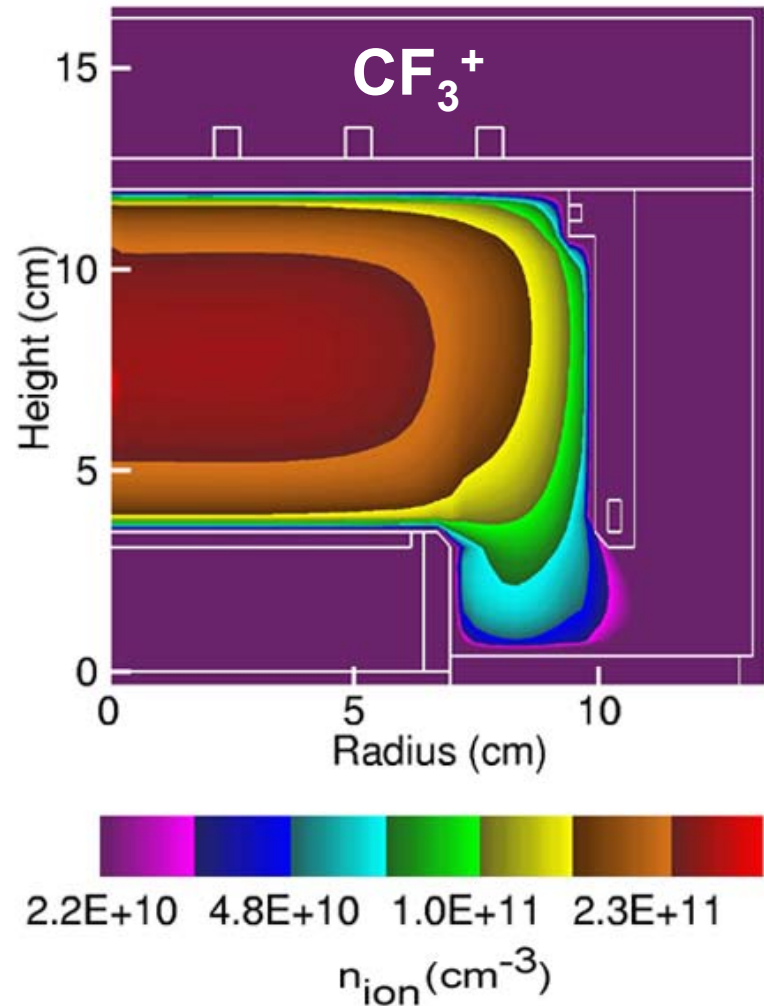


- Large power deposition typically results in near total dissociation of feedstock gases.

- C<sub>4</sub>F<sub>8</sub>, 10 mTorr, 1.4 kW, 13.56 MHz

# MAJOR POSITIVE IONS

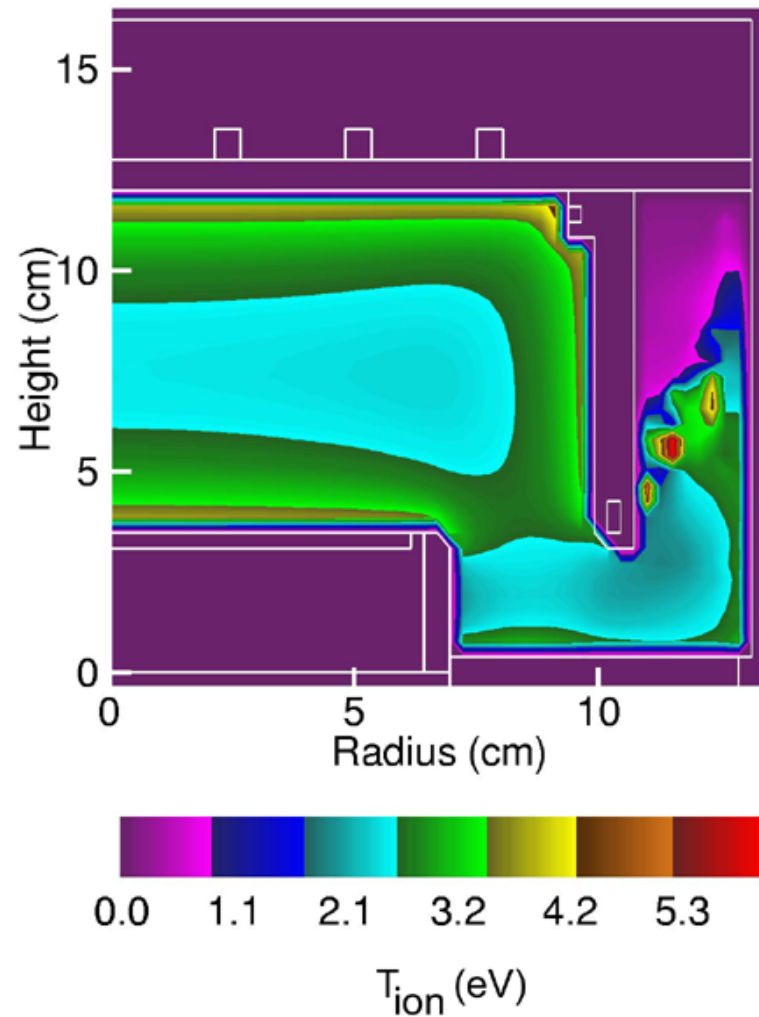
- $\text{CF}_3^+$ ,  $\text{CF}_2^+$ , and  $\text{CF}^+$  are dominant ions due to dissociation of  $\text{C}_4\text{F}_8$ .



- $\text{C}_4\text{F}_8$ , 10 mTorr, 1.4 kW, 13.56 MHz

# $\text{CF}_3^+$ TEMPERATURE

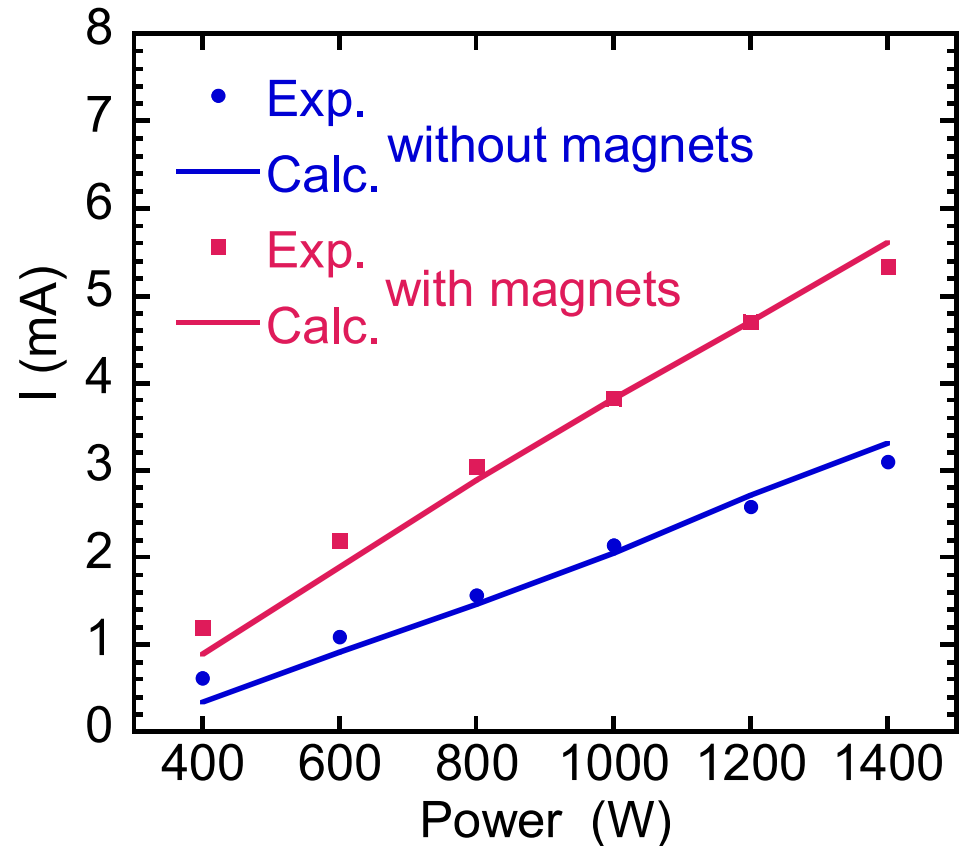
- The ion temperature is peaked near the walls where ions gain energy during acceleration in the presheath.



- $\text{C}_4\text{F}_8$ , 10 mTorr, 1.4 kW, 13.56 MHz

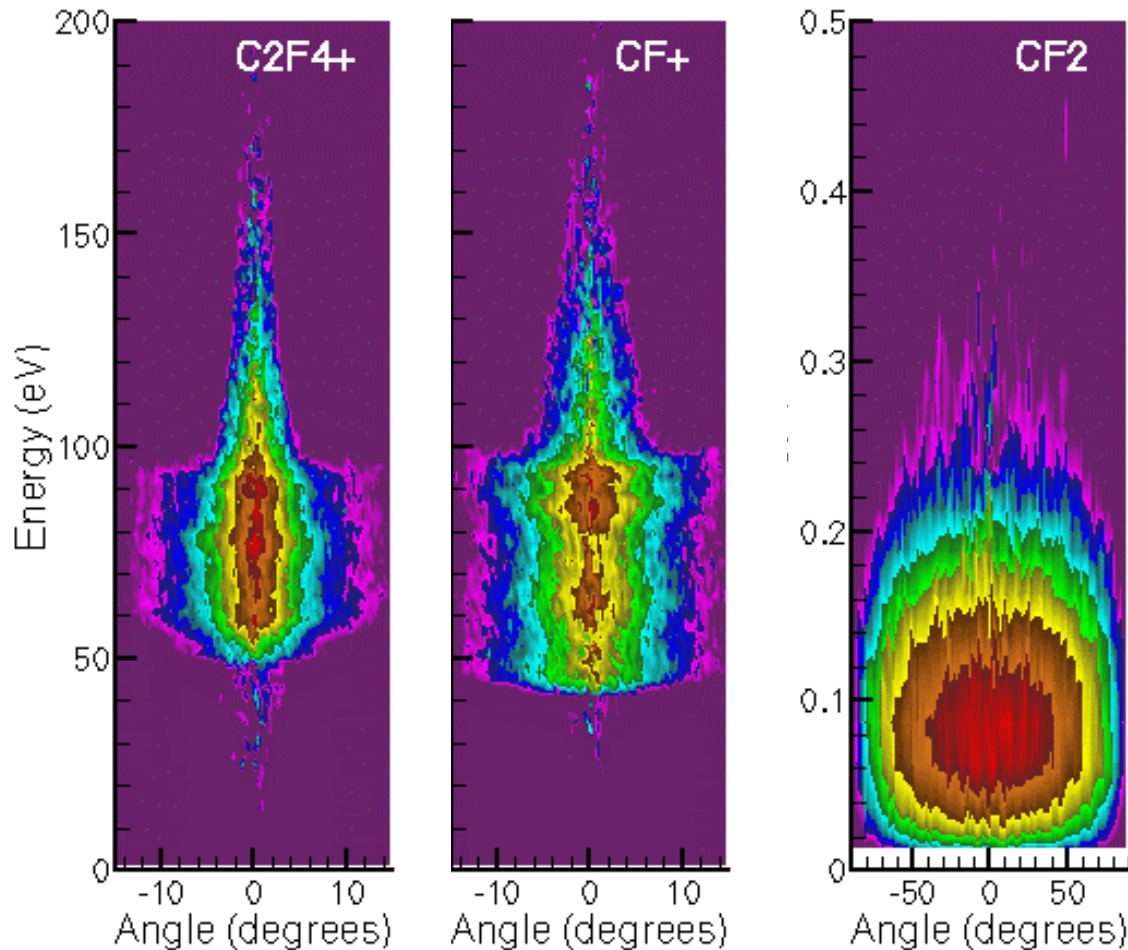
# $I_p$ VERSUS ICP POWER for $C_4F_8$

- Extensive validation of the plasma models are performed with available data for densities, temperatures and fluxes.
- Ion saturation current derived from the model are compared to experiments: ion densities are larger with moderate static magnetic fields.
- $C_4F_8$ , 10 mTorr, 13.56 MHz, 100 V probe bias
- Experiments: G. Oehrlein, Private Comm.



# ION/NEUTRAL ENERGY/ANGULAR DISTRIBUTIONS

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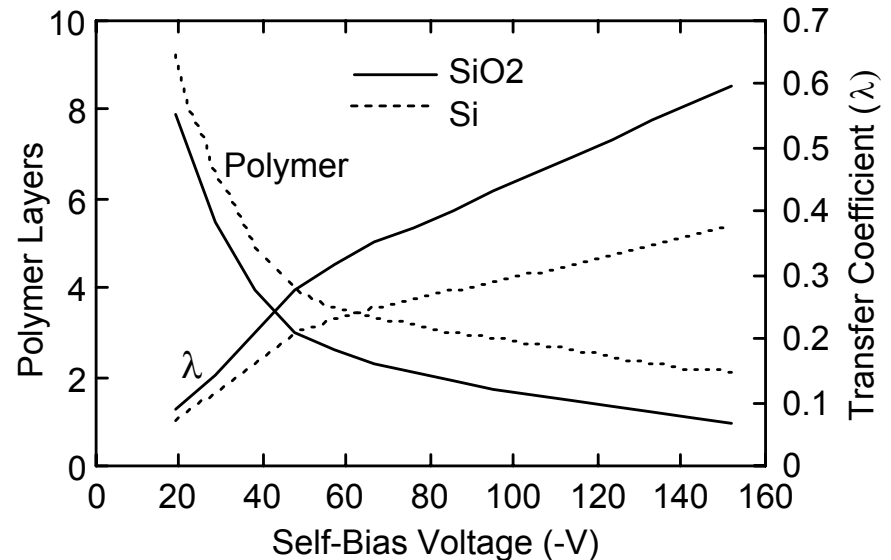
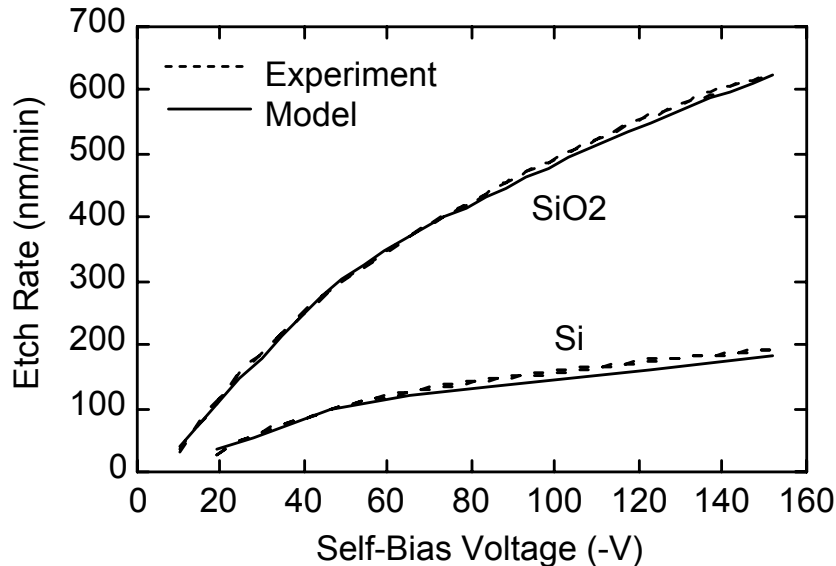


- The end products of reactor scale modeling are energy and ion angular distributions to the surface.
- In complex gas mixtures the IEADs can significantly vary from species to species.

- Ar/C<sub>4</sub>F<sub>8</sub>, 40 mTorr, 10b MHz, MERIE

# ETCH RATES AND POLYMER THICKNESS

- Etch rates for Si and SiO<sub>2</sub> increase with increasing bias due, in part, to a decrease in polymer thickness.
- The polymer is thinner with SiO<sub>2</sub> due to its consumption during etching, allowing for more efficient energy transfer through the layer and more rapid etching.

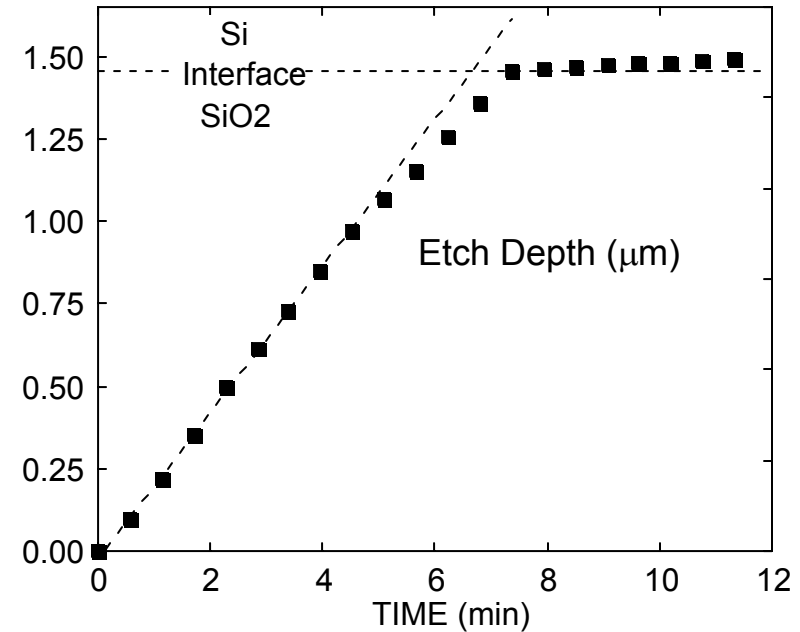
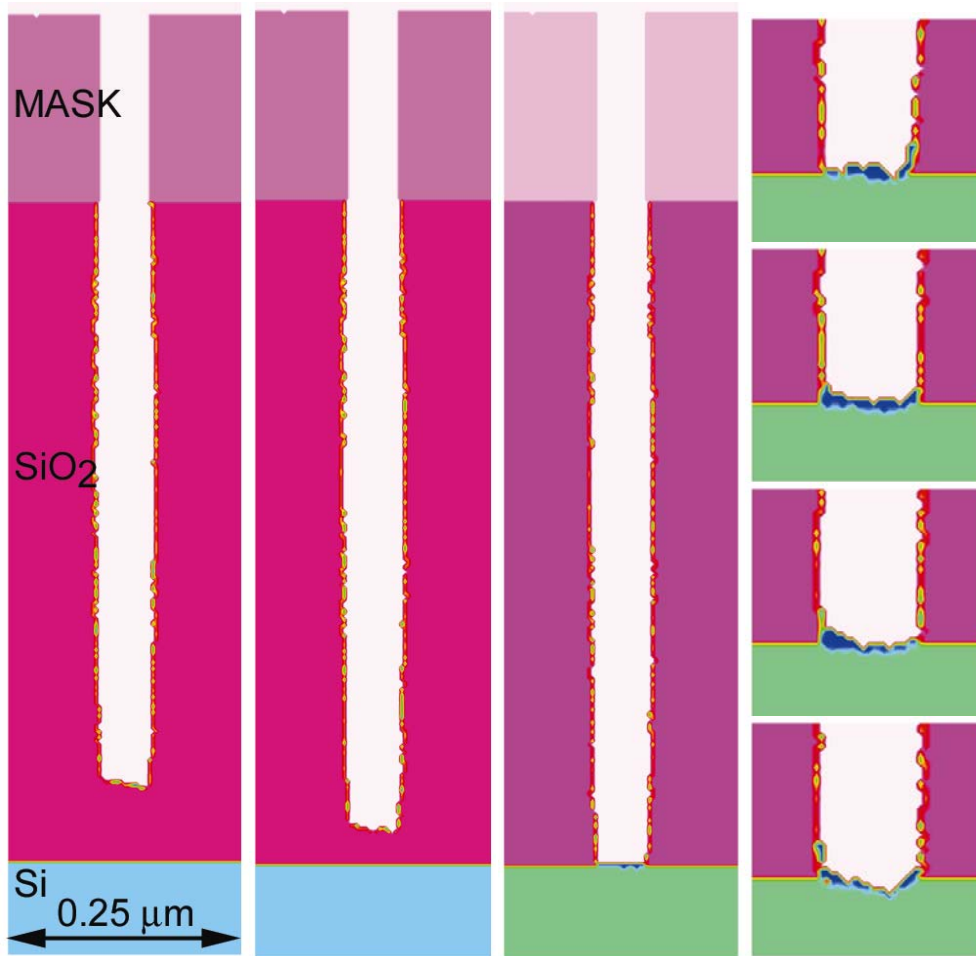


- **C<sub>2</sub>F<sub>6</sub>, 6 mTorr, 1400 W ICP, 40 sccm**

- **Exp. Ref: T. Standaert, et al.  
J. Vac. Sci. Technol. A 16, 239 (1998).**

# POLYMERIZATION AIDS SELECTIVITY

- Less consumption of polymer on Si relative to SiO<sub>2</sub> slows and, in some cases, terminates etching, providing high selectivity.

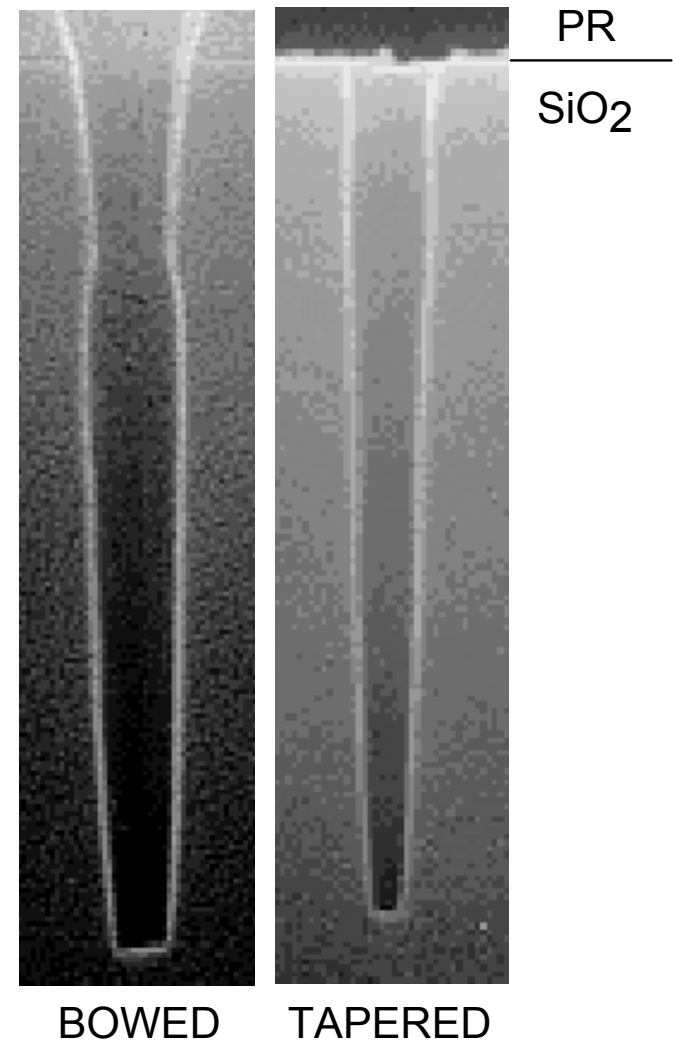




# TAPERED AND BOWED PROFILES

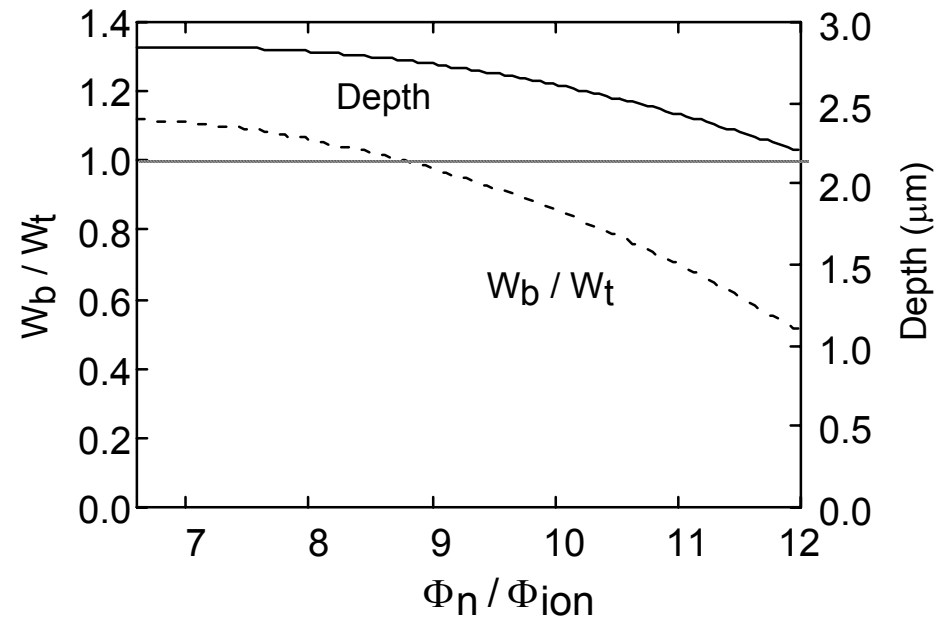
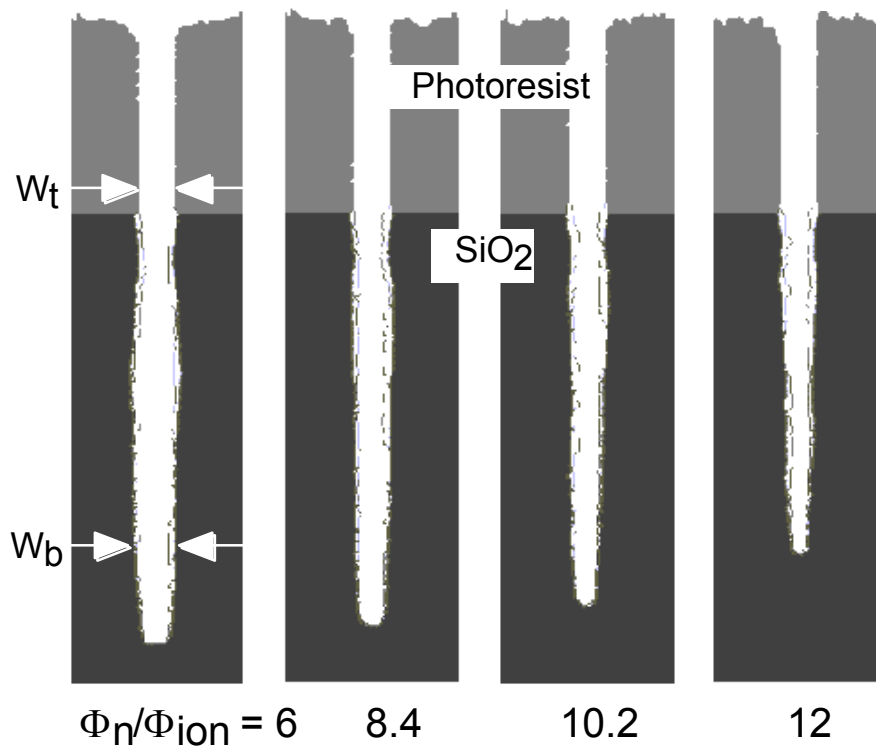
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- In high aspect ratio (HAR) etching of  $\text{SiO}_2$  the sidewall of trenches are passivated by neutrals ( $\text{CF}_x$ ,  $x \leq 2$ ) due to the broad angular distributions of neutral fluxes.
- Either tapered or bowed profiles can result from a non-optimum combination of processing parameters including:
  - Degree of passivation
  - Ion energy distribution
  - Radical/ion flux composition.



# PROFILE TOPOLOGY: NEUTRAL TO ION FLUX RATIO

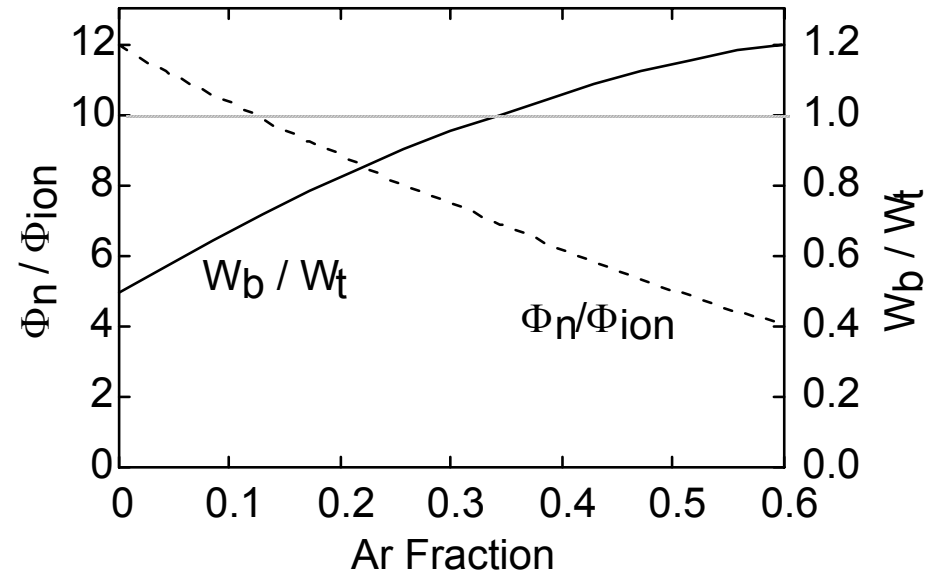
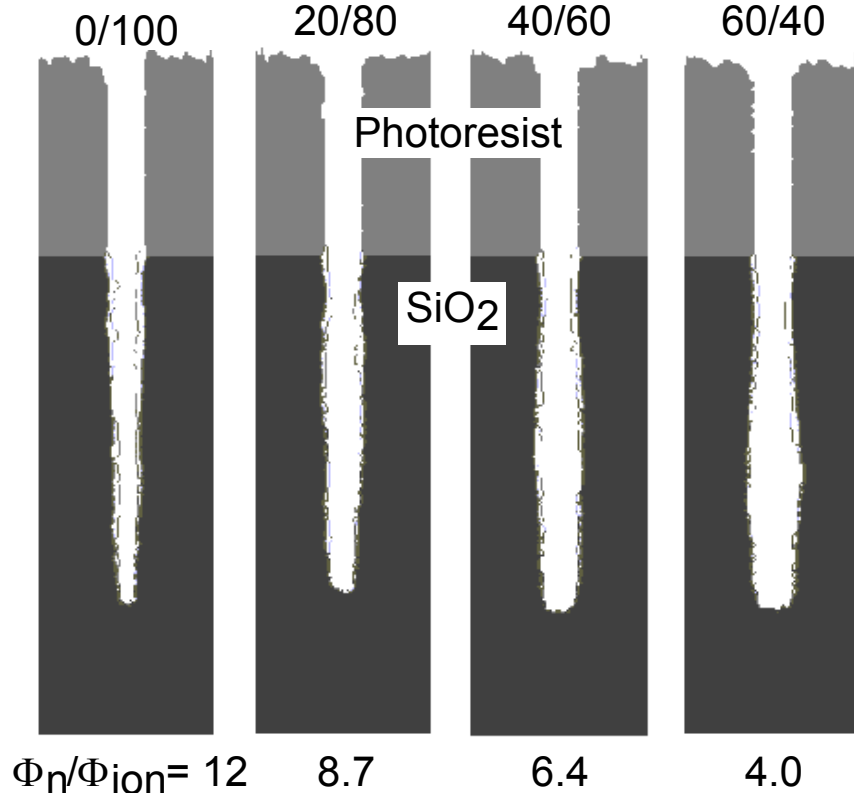
- The etch profile is sensitive to the ratio of polymer forming fluxes to energy activating fluxes. Small ratios result in bowing, large ratios tapering.



# PROFILE TOPOLOGY: ENGINEERING SOLUTIONS

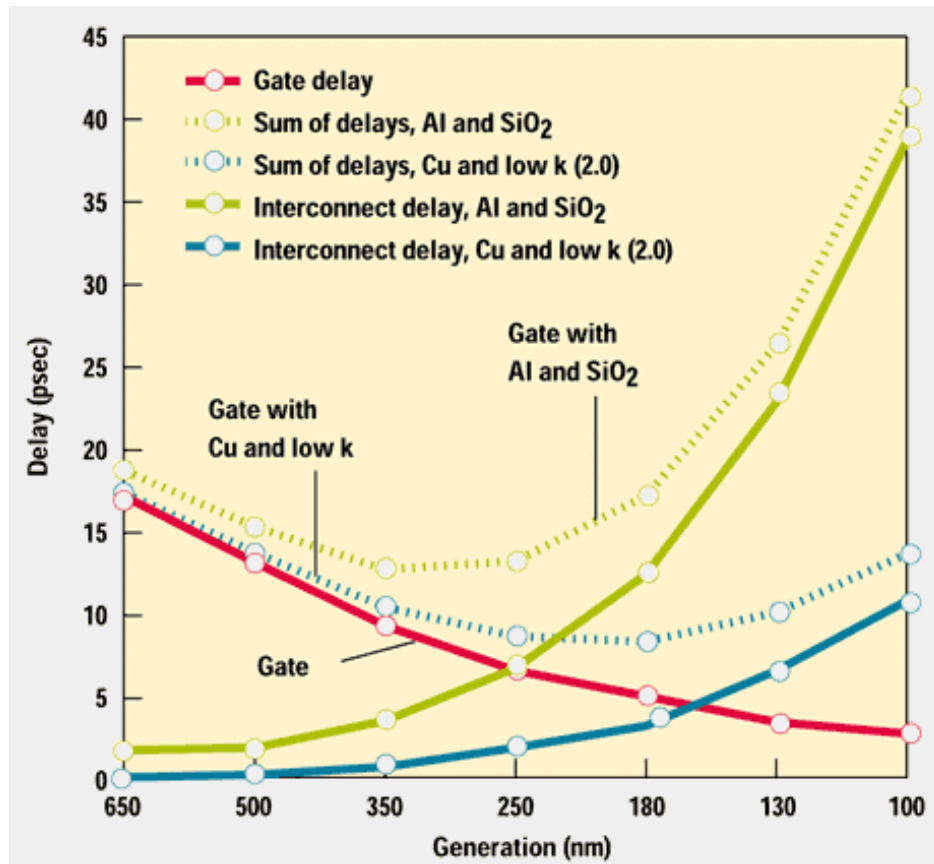
- Knowledge of the fundamental scaling parameter for controlling sidewall slop enables engineering solutions and real-time-control options.
- Example: Ar/C<sub>2</sub>F<sub>6</sub> ratio controls polymerizing/ion flux ratio, and hence profile topology.

Ar/C<sub>2</sub>F<sub>6</sub> = □



# LOW-K DIELECTRICS

- As feature sizes decrease and device count increases, the diameter of interconnect wires shrinks and path length increases.



- Large RC-delay limits processor performance.
- To reduce RC-delay, low dielectric constant (low-k) materials are being investigated.

• L. Peters, *Semi. Intl.*, 9/1/1998

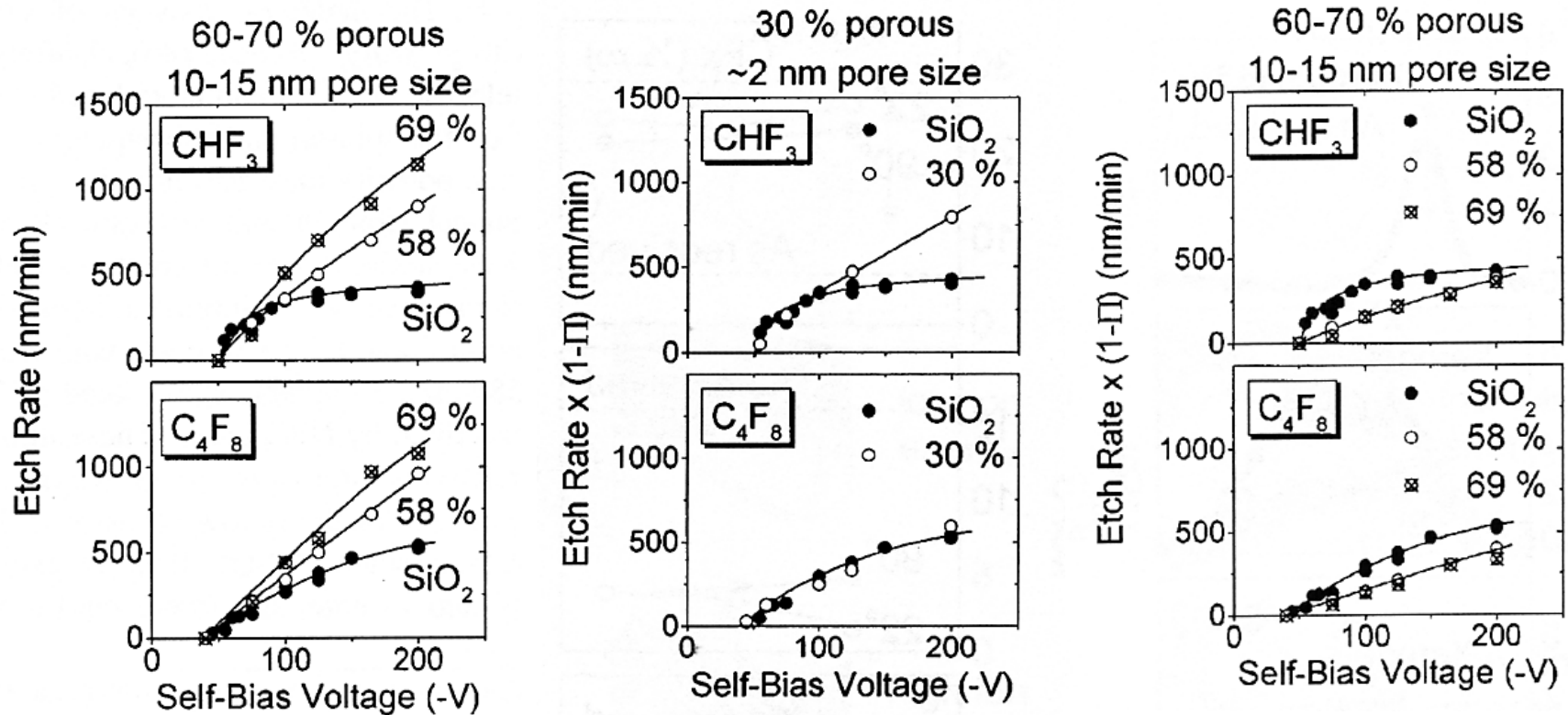
# POROUS SILICON DIOXIDE

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- Porous SiO<sub>2</sub> (xerogels) have low-k properties due to their lower mass density resulting from (vacuum) pores.
  - Typical porosities: 30-70%
  - Typical pore sizes: 2-20 nm
- Porous SiO<sub>2</sub> (P-SiO<sub>2</sub>) is, from a process development viewpoint, an ideal low-k dielectric.
  - Extensive knowledge base for fluorocarbon etching of conventional non-porous (NP-SiO<sub>2</sub>).
  - No new materials (though most P-SiO<sub>2</sub> contains some residual organics)
  - Few new integration requirements

# ETCHING OF P-SiO<sub>2</sub>: GENERAL TRENDS

- Etching of Porous SiO<sub>2</sub> typically proceeds at a higher rate than NP-SiO<sub>2</sub> for the same conditions due to the lower mass density.



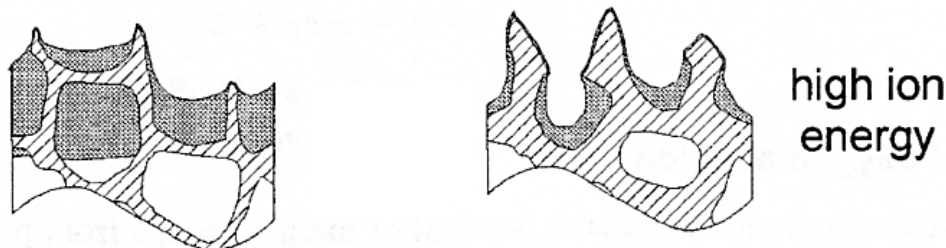
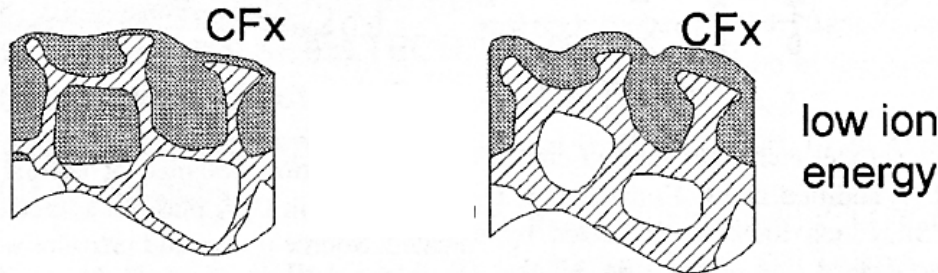
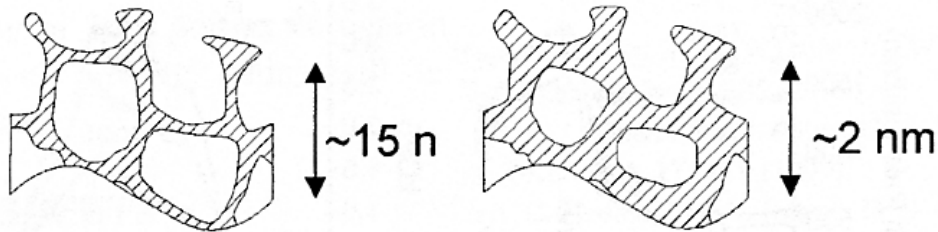
- When correcting for mass, etch rates are either larger or smaller than NP-SiO<sub>2</sub>, depending on porosity, pore size, polymerization.

- Standaert et al, JVSTA 18, 2742 (2000).

# WHAT CHANGES WITH POROUS SiO<sub>2</sub>?

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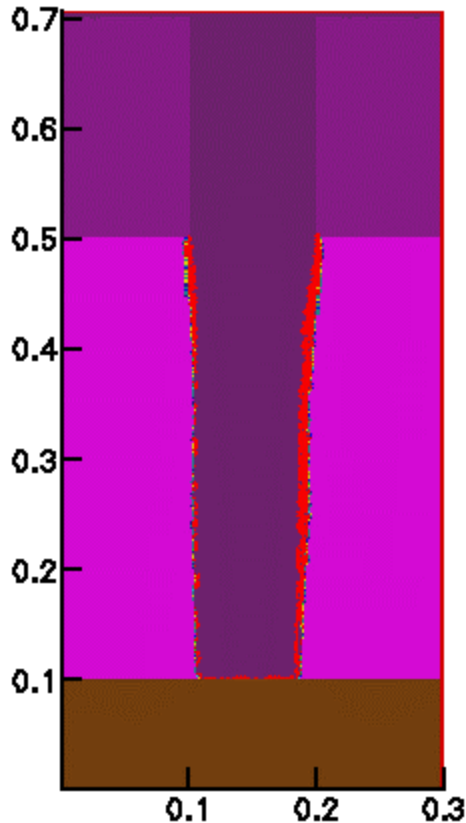
- The “opening” of pores during etching of P-SiO<sub>2</sub> results in the filling of the voids with polymer, creating thicker layers.
- Ions which would have otherwise hit at grazing or normal angle now intersect with more optimum angle.



- An important parameter is  $L/a$  (polymer thickness / pore radius).

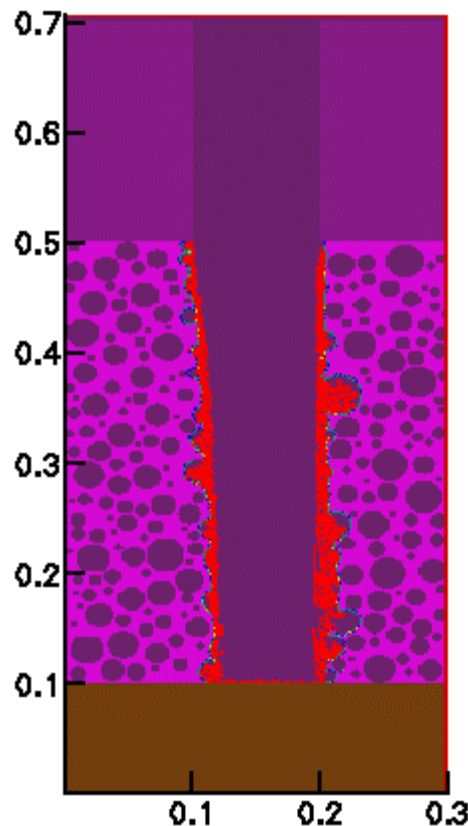
- Adapted: Standaert, JVSTA 18, 2742 (2000)

# ETCH PROFILES IN SOLID AND POROUS $\text{SiO}_2$



• Position ( $\mu\text{m}$ )

• Solid



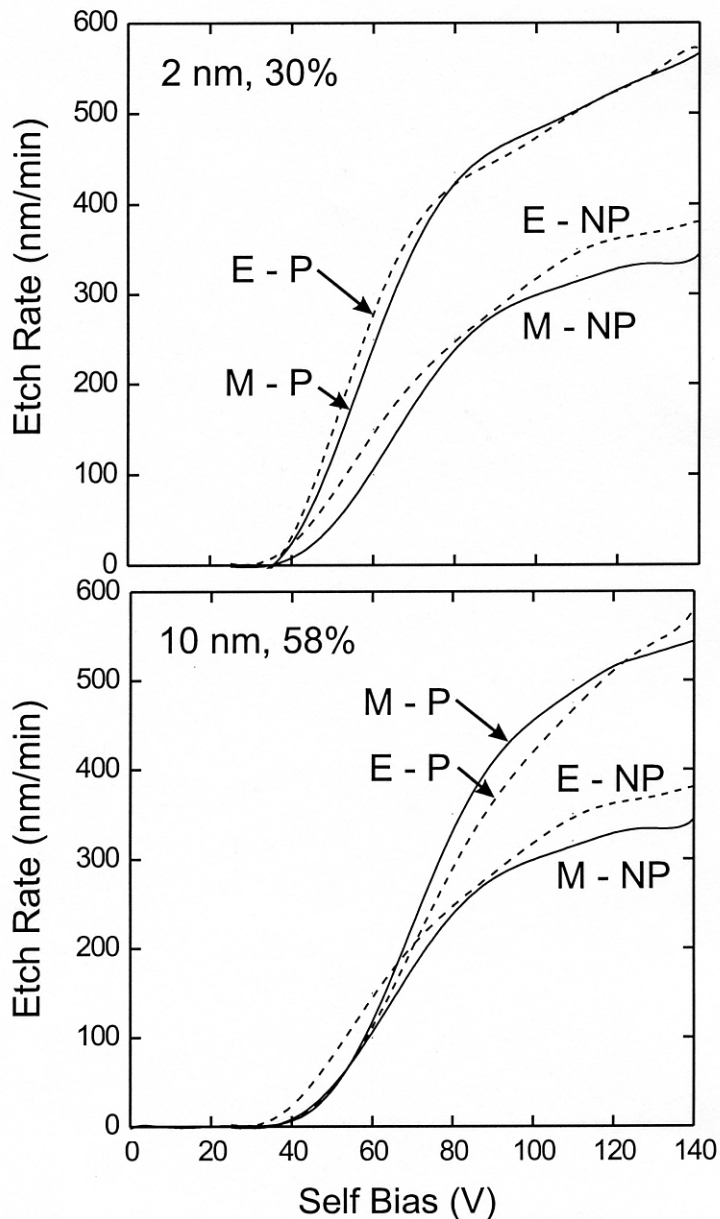
• Position ( $\mu\text{m}$ )

• Porosity = 45 %  
Pore radius = 10 nm

- Porous  $\text{SiO}_2$  is being investigated for low-permittivity dielectrics for interconnect wiring.
- In polymerizing environments with heavy sidewall passivation, etch profiles differ little between solid and porous silica.
- The “open” sidewall pores quickly fill with polymer.



# ETCHING OF POROUS SiO<sub>2</sub>



- Etch rates of P-SiO<sub>2</sub> are generally higher than for non-porous (NP).

- Examples:

- 2 nm pore, 30% porosity
- 10 nm pore, 58% porosity

- Higher etch rates are attributed to lower mass density of P-SiO<sub>2</sub>.

- CHF<sub>3</sub> 10 mTorr, 1400 W

P - Porous  
NP - Non porous  
E - Experimental  
M - Model

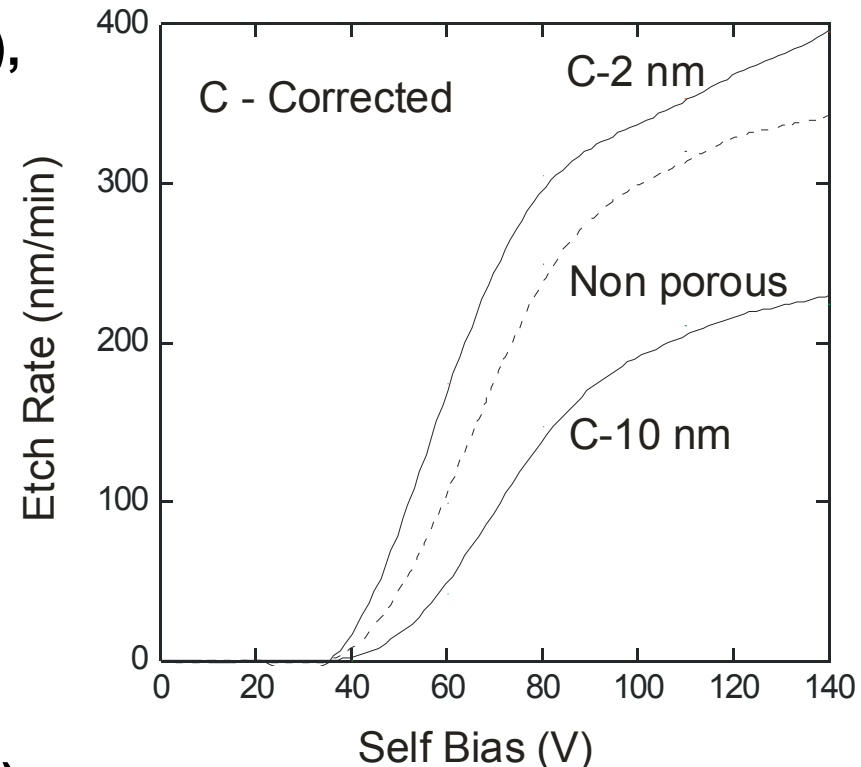
# PORE-DEPENDENT ETCHING

- To isolate the effect of pores on etch rate, corrected etch rate is defined as

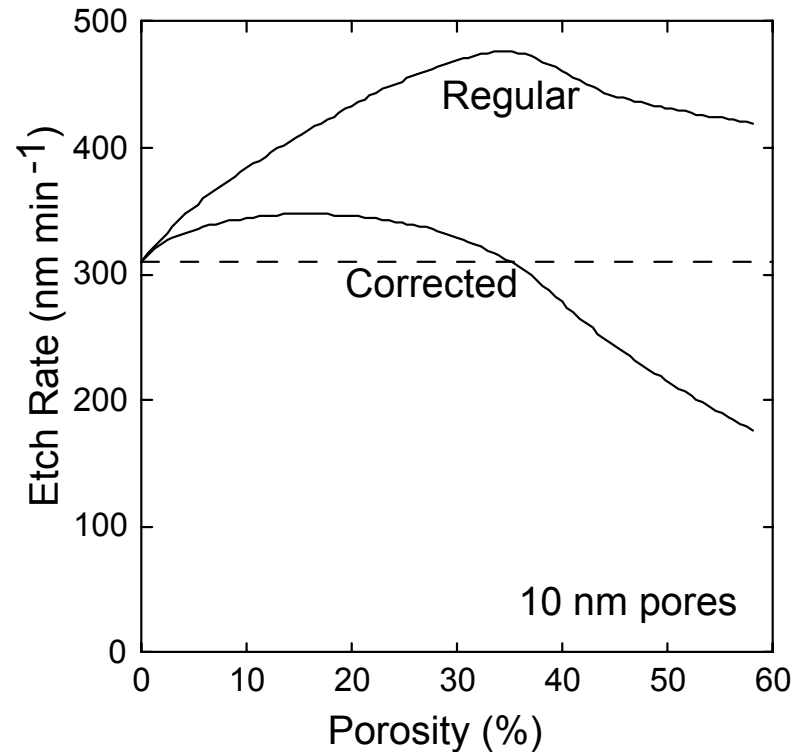
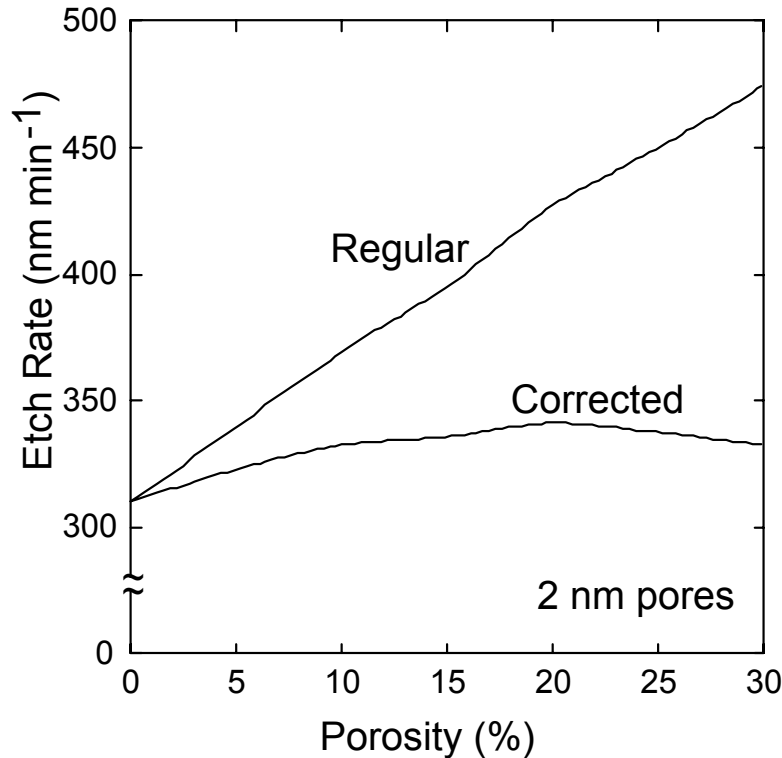
$$\text{Etch Rate (ER)}_{\text{corrected}} = \text{ER}_{\text{regular}} \times (1 - p),$$

$p$  = porosity

- If etching depended only on mass density, corrected etch rates would equal that of NP-  $\text{SiO}_2$ .
- 2 nm pores  $L/a \geq 1$  : C-ER > ER( $\text{SiO}_2$ ). Favorable yields due to non-normal incidence may increase rate.
- 10 nm pores  $L/a \leq 1$  : C-ER < ER( $\text{SiO}_2$ ). Filling of pores with polymer decrease rates.

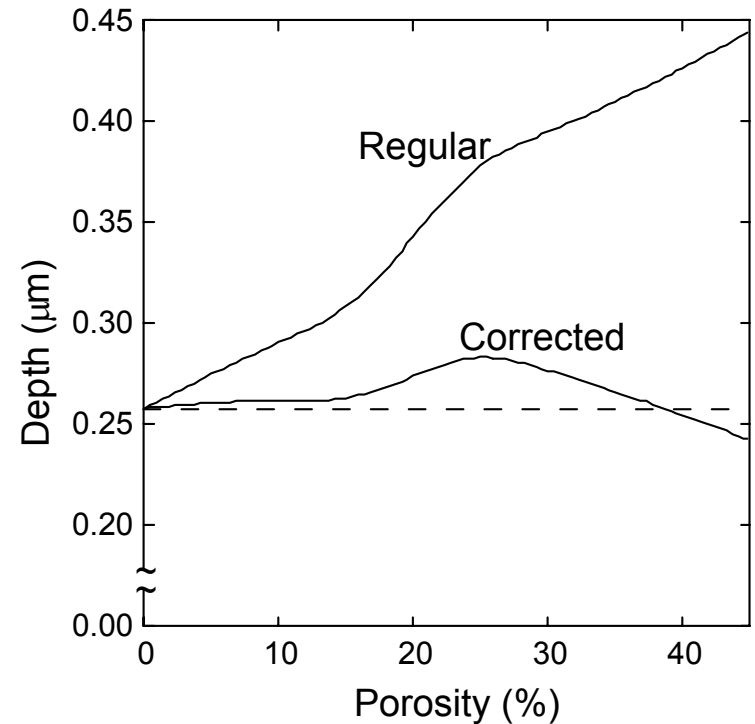
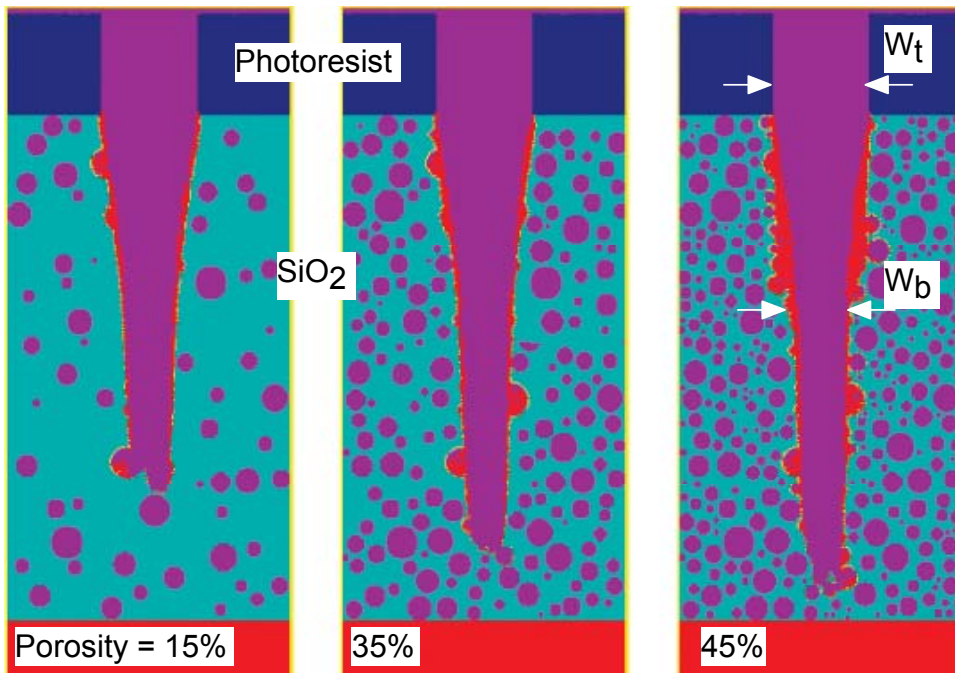


# EFFECT OF POROSITY ON BLANKET ETCH RATES



- **2 nm pores: Etch rate increases with porosity.**
- **10 nm pores: Polymer filling of pores reduces etch rate at large porosities.**

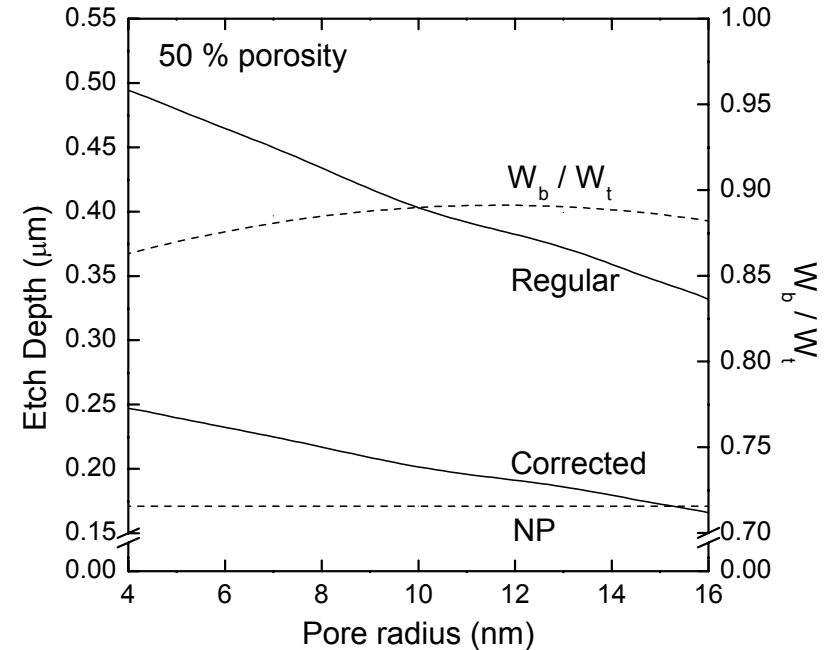
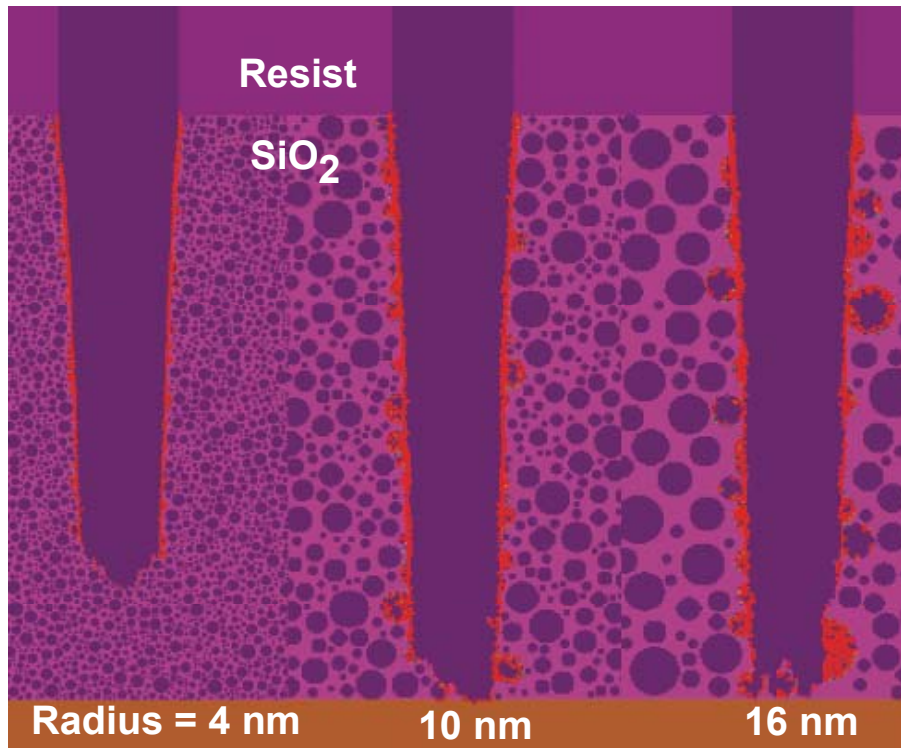
# EFFECT OF POROSITY ON HAR TRENCHES



• 10 nm pores.  $W_t = 0.1 \mu\text{m}$

- At higher porosities, more opportunity for pore filling produces thicker average polymer layers and lower etch rates.
- Corrected etch rates fall below  $\text{SiO}_2$  rates when critically thick polymer layers are formed.

# EFFECT OF PORE RADIUS ON HAR TRENCHES



- With increase in pore radius,  $L/a$  decreases, enabling pore filling and a decrease in etch rates.
- Thick polymer layers eventually leads to etch rates falling below NP. There is little variation in the taper.

# OXYGEN ETCHING OF PTFE

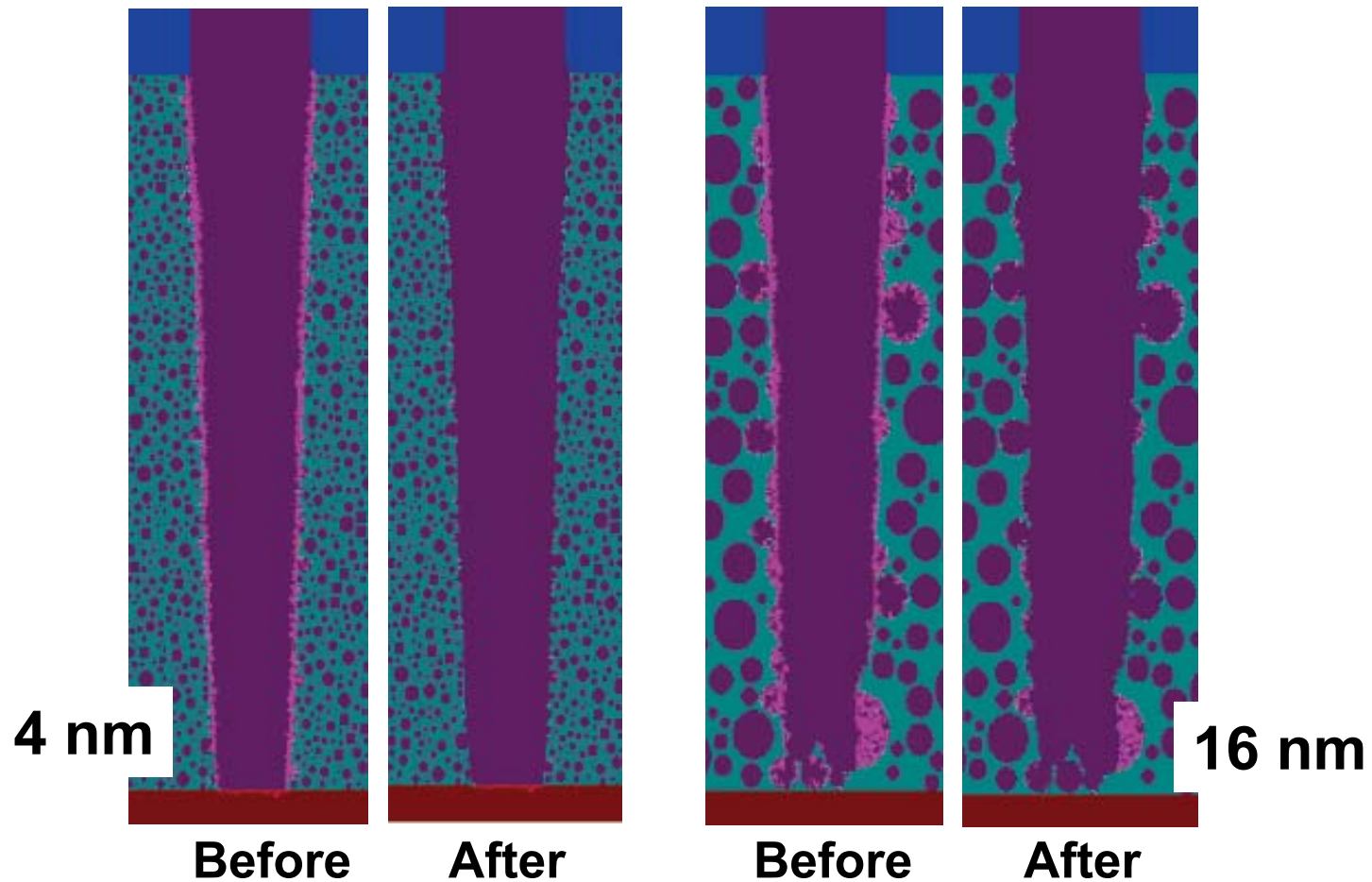
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- After etching, the polymer must be removed from the feature.
- O<sub>2</sub> plasmas are typically used for polymer stripping, usually during photoresist mask removal.
- Unlike hydrocarbon polymers which spontaneously react with O, fluorocarbon polymers require ion activation for etching.
  - Polymer + Energetic Ion → Activated Polymer Site (P\*)
  - P\* + O → Volatile Products
- Removal of polymer from porous materials is difficult due to shadowing of ion fluxes caused by the pore morphology.

# EFFECT OF PORE RADIUS ON CLEANING

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- Larger pores are more difficult to clean due small view angle of ion fluxes producing lower fluxes of less energetic ions.



# CONCLUDING REMARKS

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- **Plasmas and polymers enjoy a unique relationship in the realm of gas-surface interactions.**
- **The ability for plasmas to produce reactive species and polymers to “accept” those species at ambient temperatures have enabled a wide range of technological applications.**
- **In spite of years of use, lack of fundamental understanding of many of the basic plasma-surface processes has largely limited the technology to empirical development.**
- **As these processes become better characterized, new technologies will come to the forefront; from biocompatible surfaces to flexible display panels.**



# ACKNOWLEDGEMENTS

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- **Dr. Alex V. Vasenkov**
- **Mr. Rajesh Dorai**
- **Mr. Arvind Sankaran**
- **Mr. Pramod Subramonium**
  
- **Funding Agencies:**
  - **3M Corporation**
  - **Ford Motor Corporation**
  - **Semiconductor Research Corporation**
  - **National Science Foundation**
  - **SEMATECH**